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M8540/248465

PTO/SB/05 (8-00)

Attorney Docket No.

	APPLICATION NSMITTAL		ed Inventor tion Identifier	Gary Anthony Jubb		
(Only for new nonp	provisional applications under	Title	Sa	aline Soluble Ir	norganic Fibro	es
37	CFR 1.53(b))	Exp	express Mail Label No. EL568948202U			
	ICATION ELEMENTS cerning utility patent application contents		ADDRESS 1	Во	ssistant Director for Pa ox Patent Application /ashington, D.C 2023	atents
1. Fee Transm (Submit an original Submit an origina	mittal Form (e.g. PTO/SB/17 ginal, and a duplicate for fee processing) claims small entity status. R 1.27 In Total Pages gement as set forth below) tive title of the Invention eferences to Related Application Regarding Fed sponsore ce to sequence listing, a table of the Invention many of the Invention scription of the Drawings (if I Description I Description I Description I Total Pages evaluation [Total Pages e	48 cations ed R & D ble, or a  f filed)  (37 CFR cional with (38) (37 CFR cional with (37 CFR.)	7. CD-R Comp 8. Nucleotide (if applicable, a. Co b. Spr i. ii. c. S  ACCC 9. Assig 10. 37 C (whe 11. Engli 12. Inform State 13. Prelin 14. Return (Shout 15. Certific (If for 16. Othe Appointment Transmit Substitute Co and supply the rec 6: con-in-part (CIP)	OM or CD-R in puter Program (A e and/or Amino all necessary) imputer Readab ecification Sequal CD-ROM or Caracter Program (A e and A e	ox Patent Application lashington, D.C. 2023' duplicate, larg Appendix) Acid Sequence Die Copy (CRF) uence Listing of CD-R (2 copies) ing identity of a APPLICATION (cover sheet & tement Possignee) Document (if a ure Copies) To-1449 Coment teard (MPEP 50 all pitemized) ority Document claimed) Ittorney Tormal Dration and Power	re table or e Submission  on: ); or above copies  PARTS document(s)) ower of Attorney opplicable) pies of IDS ditations  o3) t(s)  rawings er of Attorney  preliminary
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Address	1100 Peachtree Street	1 LLP		<del></del>		
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Name (Print/Type)	Bruce D. Gray		3 (Spain 1)		Reg. No.	35,799
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FEE TRANSMITTAL	Complete if Known					
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Patent fees are subject to annual revision	Filing Date	October 24, 2000				

Small Entity payments must be supported by a small entity statement, otherwise, large entity fees must be paid. See 37 C.F.R. §§ 1.27 and 1.28

TOTAL AMOUNT OF PAYMENT (\$)710.00

Complete if Known					
Application Number	Divisional of 09/262,378				
Filing Date	October 24, 2000	Δ. S.			
First Named Inventor	Gary Anthony Jubb	.3			
Group / Art Unit	1755	- G			
Examiner Name	Group, K.	9			
Attorney Docket Number	M8540/248465	92			

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Applicant claims small entity status. See 37 CFR 1.27	113	1,840*	113	1,840*	Requesting public Examiner action	ation of SIR after	
2. Payment Enclosed:	115	110	215	55	Extension for reply	within first month	
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1. BASIC FILING FEE	118	1,360	218	680	Extension for reply	within fourth month	
Large Entity Small Entity	128	1,850	228	925	Extension for reply	within fifth month	
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101 710 201 355 Utility filing fee 710.00	120	300	220	150	Filing a brief in sup	pport of an appeal	
106 310 206 155 Design filing fee	121 138	260 1,510	221 138	130 1,510	Request for oral he Petition to institute hearing		
107 480 207 240 Plant filing fee	140	110	240	55	Petition to revive -	unavoidable	
108 690 208 345 Reissue filing fee	141	1,210	241	605	Petition to revive -		-
114 150 214 75 Provisional filing fee	142	1,210	242	605	Utility issue fee (or		-
SUBTOTAL (1) (\$)710.00	143	430	243	215	Design issue fee		
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Claims below	122	130	120	130	Petitions to the Co	mmissioner	
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** or number previously paid, if greater, For Reissues, see below  Large Entity Small Entity	581	40	581	40	Recording each pa property (times nur	tent assignment per mber of properties)	
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104 260 204 130 Multiple dependent claim	179	690	279	345	Request for Continu	`"	
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Name Kilpatrick Stockton LLP		Custo	mer N	lo. 23370	0 Telephone	(404) 815-6218	
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### IN THE UNITED STATES PATENT AND TRADEMARK OFFICE

Applicants:

Gary Anthony Jubb and Jean-Louis Martin

Serial No.:

Divisional of 09/262,378

Filing Date:

October 24, 2000

For:

SALINE SOLUBLE INORGANIC FIBRES

Box Patent Application Director of Patents and Trademarks Washington, D.C. 20231

Date: October 24, 2000

#### PRELIMINARY AMENDMENT

Sir:

Preliminary to any examination on the merits, Applicants respectfully submit the following amendments and remarks:

#### In the Specification

Kindly add the following paragraph immediately preceding line 1:

--This application is a divisional of U.S. Serial No. 09/262,378 filed March 4, 1999, now allowed, which is a continuation of U.S. Serial No. 08/899,005 filed July 23, 1997, now U.S. Patent No. 5,994,247 issued on November 30, 1999, which is a continuation of U.S. Serial No. 08/535,587 filed September 28, 1995, now abandoned, which is a continuation of U.S. Serial No. 08/039,086, filed April 9, 1993, now abandoned, which claims priority to International Application No. PCT/GB93/00085, filed January 15, 1993, which claims priority to GB 92 00993.5, filed January 17, 1992 and GB 92 24612.3, filed November 24, 1992--.

#### In the Claims

Please cancel claims 1-13 without prejudice or disclaimer to the subject matter thereof.

Please add the following new claims.

Express Mail Label No. EL568948202US Divisional of U.S. Serial No. 09/262,378 "Saline Soluble Inorganic Fibers" Filed: October 24, 2000

PRELIMINARY AMENDMENT

--14. A method of insulating an article against temperatures which may on occasion exceed 900°C comprising:

disposing on, in, near or around the article thermal insulation which is a refractory insulating material having a maximum service temperature greater than 900°C and comprising vitreous fibers having a composition comprising SiO<sub>2</sub>, CaO, MgO, and optionally Al<sub>2</sub>O<sub>3</sub>, wherein:

- (a)  $SiO_2$  is present in an amount (1) greater than 58% by weight  $SiO_2$ , if the amount of MgO in the composition is in the range 0 through 10 percent by weight; or (2) greater than the sum of (58 + 0.5) (weight percent of MgO 10) percent by weight  $SiO_2$ , if the amount of MgO in the composition is greater than 10 percent by weight;
  - (b) an amount up to 42 percent by weight CaO;
  - (c) an amount up to 31.33 percent by weight MgO., and
  - (d) 0 to less than 3.97 percent by weight Al<sub>2</sub>O<sub>3</sub>;

wherein the refractory insulation material has a maximum service temperature greater than 900°C; the refractory insulation material has a shrinkage of less than 3.5 percent when exposed to a temperature of 1000°C for 24 hours, and has a shrinkage of less than 3.5 percent when exposed to a temperature of 800°C for 24 hours; and

wherein the refractory insulation material is essentially free of alkali metal oxide and boron oxide fluxing components.--

--15. The method of claim 14, wherein the amount of  $SiO_2$  is less than 70 percent by weight. --

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- --16. The method of claim 14, wherein the vitreous fibers crystallize as wollastonite, pseudowollastonite, or a mixture thereof after exposure to a temperature of 1000°C, and wherein said wollastonite, pseudowollastonite, or mixture thereof comprises:
  - (a) 60 through 67 percent by weight SiO<sub>2</sub>;
  - (b) 26 through 35 percent by weight CaO;
  - (c) 4 through 6 percent by weight MgO; and
  - (d) 0 through 3.5 percent by weight  $Al_2O_3$ . --
  - --17. The method of claim 14, wherein the vitreous fibers comprise:
    - (a) a maximum of 71.24 percent by weight SiO<sub>2</sub>;
    - (b) 4.46 through 34.49 percent by weight CaO;
    - (c) 1.71 through 22.31 percent by weight MgO; and
    - (d) 0 through 2.57 percent by weight  $Al_2O_3$ . --
  - --18. The method of claim 17, wherein the vitreous fibers further comprise:
    - (e) 0 through 0.65 percent by weight Na<sub>2</sub>O;
    - (f) 0 through 0.13 percent by weight  $K_2O$ ;
    - (g) 0.08 through 0.4 percent by weight  $Fe_2O_3$ ; and
    - (h) 0 through 1.23 percent by weight ZrO<sub>2</sub>. --
- --19. The method of claim 14, further comprising forming said vitreous fibers into a bulk, blanket, block, or vacuum-formed form.--
- --20. The method of claim 14, wherein said insulated article is occasionally exposed to a temperature above around 1000°C.--

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- --21. The method of claim 19 wherein said vitreous fibers are formed into a needled blanket.--
- --22. The method of claim 14, wherein the vitreous fibers crystallize as diopside after exposure to a temperature of 1000°C, and wherein said diopside comprises:
  - (a) 59 through 64 percent by weight SiO<sub>2</sub>;
  - (b) 19 through 23 percent by weight CaO;
  - (c) 14 through 17 percent by weight MgO; and
  - (d) 0 through 3.5 percent by weight  $Al_2O_3$ .--

#### REMARKS

New claims 14-22 are fully supported by the specification as originally filed, and correspond to claims 23-28, 30, and 31 in parent application Serial No. 09/262,378. No new matter has been added.

Applicants respectfully submit that the claims are novel and nonobvious over Olds et al. (U.S. Patent No. 5,332,699), Olds et al. (WO 87/05007), or Karppinen et al., and that the rejections made by the Examiner over these references in the parent application should not be repeated.

The claims recite a process for insulating an article against temperatures that may exceed 900 °C. None of the above references teach insulating against such a high temperature, which is

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"Saline Soluble Inorganic Fibers"

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quite different from insulating against another, lower temperature. As a result, the references do

not anticipate the claimed invention. Moreover, there is no teaching or suggestion to use the

materials described in the above references to insulate against such high temperatures. In

particular, there is no teaching or suggestion that the materials described in the references would

be able to withstand temperatures that exceed 900 °C without shrinking. As the Examiner is no

doubt aware, even if a fiber can withstand high temperatures, if it shrinks substantially when

exposed to those temperatures, then its value as insulation is very low, since the shrinkage will

cause gaps in the insulation.

Applicants respectfully submit that the claims are in condition for allowance, and an early

notification to that effect is earnestly solicited.

Please charge any additional fees or credit any overpayment to Deposit Order Account

No. 11-0855.

Respectfully submitted,

Bruce D. Gray

Reg. No. 35,799

Attorney for Assignee

OF COUNSEL:

KILPATRICK STOCKTON LLP

**Suite 2800** 

1100 Peachtree Street

Atlanta, Georgia, 30309-4530

404-815-6218

Attorney Docket No.: M8540/248465

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#### SALINE SOLUBLE INORGANIC FIRRES

This invention relates to saline soluble. non-metallic, amorphous, inorganic oxide, refractory fibrous materials.

Inorganic fibrous materials are well known and widely used for many purposes (e.g. as thermal or acoustic insulation in bulk, mat, or blanket form, as vacuum-formed shapes, as vacuum formed boards and papers, and as ropes, yarns or textiles; as a reinforcing fibre for building materials; as a constituent of brake blocks for vehicles). In most of these applications the properties for which inorganic fibrous materials are used require resistance to heat, and often resistance to aggressive chemical environments.

Inorganic fibrous materials can be either glassy or crystalline. Asbestos is an inorganic fibrous material one form of which has been strongly implicated in respiratory disease.

It is still not clear what the causative mechanism is that relates some asbestos with disease but some researchers believe that the mechanism is mechanical and size related. Asbestos of a critical size can pierce cells in the body and so, through long and repeated cell injury, have a bad effect on health.

whether this mechanism is true or not regulatory agencies have indicated a desire to categorise any inorganic fibre product that has a respiratory fraction as 'hazardous', regardless of whether there is any evidence to support such categorisation. Unfortunately for many of the applications for which inorganic fibres are used, there are no realistic substitutes.

Accordingly there is a demand for inorganic fibres that will pose as little risk as possible (if any) and for which there are objective grounds to believe them safe.

A line of study has proposed that if inorganic fibres were made that were sufficiently soluble in physiological fluids that their residence time in the human body was short; then damage would not occur or at least be minimised. As the risk of asbestos-linked disease appears to depend very much on the length of exposure this idea appears reasonable. Asbestos is extremely insoluble.

As intercellular fluid is saline in nature the importance of fibre solubility in saline solution has long been recognised. If fibres are soluble in physiological saline solution then, provided the dissolved components are not toxic, the fibres should be safer than fibres which are not so soluble. The shorter the time a fibre is resident in the body the less damage it can do. H. Förster in 'The behaviour of mineral fibres in physiological solutions' (Proceedings of 1982 WHO IARC Conference, Copenhagen, Volume 2, pages 27-55(1988)) discussed the behaviour of commercially produced mineral fibres in physiological saline solutions. Fibres of widely varying solubility were discussed.

International Patent Application No. W087/05007 disclosed that fibres comprising magnesia, silica, calcia and less than 10 wt% alumina are soluble in saline solution. The solubilities of the fibres disclosed were in terms of parts per million of silicon (extracted from the silica-containing material of the fibre) present in a saline solution after 5 hours of exposure. The highest value revealed in the examples had a silicon level of 67 ppm. In contrast, and adjusted to the same regime of measurement, the highest level disclosed in the Förster paper was equivalent to approximately 1 ppm. Conversely if the highest value revealed in the International Patent Application was converted to the same measurement

regime as the Förster paper it would have an extraction rate of 901,500 mg Si/kg fibre - i.e. some 69 times higher than any of the fibres Förster tested, and the fibres that had the highest extraction rate in the Förster test were glass fibres which had high alkali contents and so would have a low melting point. This is convincingly better performance even taking into account factors such as differences in test solutions and duration of experiment.

International Patent Application No. WO89/12032 disclosed additional fibres soluble in saline solution and discusses some of the constituents that may be present in such fibres.

European Patent Application No. 0399320 disclosed glass fibres having a high physiological solubility.

Further patent specifications disclosing selection of fibres for their saline solubility are European 0412878 and 0459897, French 2662687 and 2662688, PCT WO86/04807 and WO90/02713.

The refractoriness of the fibres disclosed in these various prior art documents varies considerable. The maximum service temperature of any of these disclosed fibres (when used as refractory insulation) is up to 815°C (1500°F).

Service temperature for refractory insulation is definable in many ways but to be consistent with the above mentioned International Patent Applications this application shall mean by service temperature that temperature at which the fibre shows acceptable shrinkage (maximum of 5% linear shrinkage after exposure to temperature for 24 hours) and at which the fibre has not appreciably suffered through excessive sintering or softening.

There is a demand for physiologically soluble fibres having a service temperature of greater than 815°C,

particularly for such fibres having a service temperature above 900°C.

Testing for physiological solubility and safety can be done by inhalation studies on, e.g. rats. However such studies are extremely time consuming and costly. A study can take of the order of 2½ years from start and can easily cost f1 million per study. A cheaper alternative is to test for solubility in physiological or like fluids in vitro.

Testing of an inorganic fibre for solubility in physiological solutions is not so time consuming, but there is currently no way of predicting which systems will produce such soluble fibres. Therefore anyone seeking to find such soluble fibres has to work on a trial and error basis assisted by what is commonly known as 'chemical intuition' but is equally commonly known as 'a hunch'. Such trial and error testing is laborious and time consuming. Further, once a fibre is found that is soluble there is no guarantee that it will be usable at useful service temperatures.

Accordingly there is a demand for a method of predicting whether a fibre will have a reasonable solubility in physiological solutions, and further there is a demand that such a test should preferably give an indication as to expected service temperature.

Shrinkage of inorganic refractory fibres occurs through two mechanisms; the first is viscous flow of the fibre material. Most inorganic refractory fibres are glasses and so may be defined as liquids having an exceedingly high viscosity (but still liable to flow). By their nature fibres are elongate and so have a high surface area per unit volume. As the reduction of surface area is a means of reducing the surface energy of a material, when the glass becomes fluid enough it will flow so as to reduce surface area. This flow results in a coarsening and shortening of the fibres and so to

shrinkage, and in the extreme results in disruption of the fibres into separate particles.

The second mechanism leading to shrinkage is that at elevated temperatures glasses may crystallise to form one or more crystal phases. Usually these crystal phases have a smaller molar volume than the glasses from which they crystallise and so shrinkage results. Some fibres are known for which the molar volume of the crystalline form exceeds that of the glass (for example  ${\rm Al}_2{\rm O}_3/{\rm SiO}_2$  glassy fibres may crystallise to form mullite crystals). In these cases the expansion due to crystallisation may oppose the shrinkage caused by viscous flow.

If shrinkage through viscous flow occurs at a much lower temperature than crystallisation then the crystallisation may not be able to compensate for such shrinkage.

There is a demand for a fibre in which both viscous flow and crystallisation occur at as high and as similar a temperature as possible, and preferably in which the expansion due to crystallisation closely matches the shrinkage due to viscous flow so that the net effect is as close to zero shrinkage as possible.

When used as refractory insulation inorganic refractory fibres are used in several forms. The fibres may be supplied as a bulk material, but in this form the fibres are difficult to handle for many applications. Alternatively the fibre may be supplied as a blanket. Blanket fibre is generally made by a process of sucking fibre from air onto a conveyor to form a blanket. Because the fibres tend to be aligned parallel to the conveyor surface they can separate easily. Accordingly the blanket fibres are secured together by adding a binder to lock the fibres together, or by needling the blanket, or both. In needling needles are passed through the thickness of the blanket to push and draw fibres to lie transverse to the

blanket and so tie the fibres together. Because binders are usually resins, such as phenolic resins, they burn off on first firing. There is a desire to reduce the amount of such binders used both because of possible health implications in handling, and because the combustion products may affect the strength of the fibres. Thus needled blanket is usually preferred.

The fibres may also be supplied as blocks, generally made from assembled layers of inorganic fibre blanket.

For some fibres needling is not possible. Crystalline fibres—are generally too brittle to stand the stresses involved. For the fibres known in the industry as glass fibres (which are generally used for low temperature applications) the amount of 'shot' (unfiberised glass particles) present is generally too high to allow needling as the shot damages the needles. There is no needled blanket on the market that has a maximum service temperature in the range 900°C-1200°C. There are needled blankets having a higher maximum service temperature but these use expensive fibres in comparison with other fibres usable (with the aid of binders) as blanket in the temperature range 900°C-1200°C.

Accordingly there is a demand for needled fibre blanket formed from inexpensive materials, being soluble in saline solutions, and having a maximum service temperature in the range 900°C-1200°C.

As stated previously refractory oxide fibres are made by several methods all of which involve the formation of a melt of oxides and the subsequent fiberisation of the melt by e.g. spinning or blowing.

The melt of oxide material is often formed by electrical discharge melting of the constituent raw materials. The applicants, in manufacture of a CaO/MgO/SiO<sub>2</sub> refractory oxide

fibre encountered problems due to the necessity of handling CaO. These problems were discovered to be due to the moisture content of CaO as commercially available. One of the problems of use of CaO is the outgasing that results upon melting and this led at the least to a porous melt pool which caused fluctuations in the melt current; in the extreme the outgasing was explosive. Additionally use of CaO appeared to cause accelerated attack on the melt electrodes. Also CaO is a difficult and corrosive material to handle.

Accordingly there is a need for a process that minimises the use of CaO.

Accordingly the present invention provides the following features both independently and in combination:-

- A. Use of a vitreous inorganic fibre in the knowledge that it has a composition meeting the criteria either: that the calculated sum of the free energies of hydration of the compounds that would or could be present at equilibrium (on the basis of knowledge, informed belief or reasonable assumption) is more negative than -10 kcal/100grams (-418.4kJ/kg) of composition; or that the percentage of non-bridging oxygens is more than 30%. Such compositions tend to be saline soluble.
- B. Use of such a vitreous inorganic fibre in the knowledge that it has a composition meeting the criterion that the ratio of glass modifiers to network formers is less than a critical value (for SiO<sub>2</sub> based compositions the critical value is 1). Such compositions tend to be glass formers.
  - <u>c</u>. The invention also encompasses fibres selected by adopting such criteria as a test for solubility and glass formation.
  - D. Use as saline soluble fibres having a shrinkage of less

than 3.5% when exposed to 1000°C for 24 hours and/or 800°C for 24 hours, of vitreous fibres having a composition comprising (in weight %):-

$$SiO_2$$
 >58% - (for MgO =< 10%) and  $SiO_2$  >58% + 0.5(%MgO -10) - (for MgO >= 10%) (SiO<sub>2</sub> preferably being below 70%) CaO 0% - 42% MgO 0% - 31.33% Al<sub>2</sub>O<sub>3</sub> 0% - <3.97%

and being essentially free of fluxing components such as alkali metals and boron oxide.

E. In one such usage the first crystalline material resulting on crystallisation has the crystal structure of diopside and has the composition consisting essentially of:-

<u>Component</u>	Composition A
	Weight percent
SiO <sub>2</sub>	59-64
Al <sub>2</sub> O <sub>3</sub>	0-3.5
CaO	19-23
MgO	14-17

F. In a second such usage the first crystalline material resulting on crystallisation has the crystal structure of wollastonite/pseudowollastonite and has the composition consisting essentially of:-

Component	Composition B
	Weight percent
sio <sub>2</sub>	60-67
Al <sub>2</sub> 0 <sub>3</sub>	0-3.5
CaO	26-35
MgO	4-6

- G. The fibres used in such manner may further be used as needled blankets.
- H. Preferably the fibres of the general composition and compositions A and B mentioned above have a SiO<sub>2</sub> content (expressed as a weight percentage of the constituents SiO<sub>2</sub>, CaO and MgO) of greater than 60%.
- I. The present invention further provides a method for the manufacture of refractory oxide fibres containing calcium and silicon by the formation of an oxide melt containing calcium and silicon characterised in that all or part of the calcium and all or part of silicon is provided by a calcium silicate.

The invention is illustrated by way of example in the following description and with reference to the drawings in which:-

- Fig.1 is a three-axis phase diagram indicating the crystalline phases in the system SiO<sub>2</sub>/CaO/MgO (Phase Diagrams for Ceramists, The American Ceramic Society, 1964) a key to this diagram is at the end of the specification;
- Fig.2 is a three-axis composition plot of the projection onto the  ${\rm SiO}_2/{\rm CaO}/{\rm MgO}$  phase field of compositions comprising  ${\rm SiO}_2$ , CaO, MgO and  ${\rm Al}_2{\rm O}_3$ ;
- Fig.3 is a temperature/time plot of the firing regime used in a series of cyclic exposure tests of experimental

compositions;

Fig.4 is a plot of log (total solubility) v calculated free energy of hydration for a series of fibres.

Fig. 5 is a plot of log (total solubility) v % non-bonding oxygens for a series of fibres (see below).

A series of fibres were made of the compositions shown in Table 1. These fibres were melt spun by using a vertical spinning system of the type known for making inorganic fibres. Also shown in Table 1 are the compositions of some comparative commercially available inorganic oxide fibres and glass fibres.

TABLE 1

	Al <sub>2</sub> O <sub>3</sub>	sio <sub>2</sub>	CaO	MgO	Zro <sub>2</sub>	
SW-A SW-A1 SW-A2	1.1	59.36× 63.766 60.86+	20.5	15.5 15.2 15.4	-	
SW-B1 SW-B2 SW-B3	2.3 1.3 1.0	65.3 66.9 60.0	26.8 27.5 34.0	5.7 5.2 4.4	- - -	
	A1 <sub>2</sub> 0 <sub>3</sub>	sio <sub>2</sub>	COMPARA CaO	TIVE E	ZrO <sub>2</sub>	MST
CRBT CWBT CHBT	46.5 40.6 49.7	53 49.5 35.1	0.04 5.50 0.04	0.01 4.00 0.01	- - 14.7	1260°C 870°C 1425°C
Glass Fibre	15.2/ 15.5	53.7/ 57.5	21.1/21.8	1.3/	<b>-</b>	+ 5.9-6.2% B <sub>2</sub> O <sub>3</sub> 0.11-0.12% TiO <sub>2</sub> 0.46% Na <sub>2</sub> O 0.32-0.33% K <sub>2</sub> O
Needled Glass Fibre	3.7	60.5/ 60.0	8.1 7.9	4.0		+ 2.85-2.95% B <sub>2</sub> O <sub>3</sub> 13.5% Na <sub>2</sub> O 1.0% K <sub>2</sub> O

[MST = Maximum Service Temperature (oxidising atmosphere)]

The fibres SW-A, SW-A1, SW-A2, SW-B1, SW-B2 and SW-B3 were tested for solubility by the following method.

The fibre was first chopped in the following manner. 2.5 g of fibre (deshotted by hand) was liquidised with 250 cm $^3$  of distilled water in a domestic Moulinex (Trade Mark) food blender for 20 seconds. The suspension was then transferred to a 500 cm $^3$  plastic beaker and allowed to settle after which as much liquid as possible was decanted and the remaining liquid removed by drying in an oven at 110°C.

The solubility test apparatus comprised a shaking incubator water bath, and the test solution had the following composition:-

Compound	. Name	Grams
NaCl	Sodium chloride	6.780
NH <sub>4</sub> Cl	Ammonium chloride	0.540
NaHCO <sub>3</sub>	Sodium bicarbonate	2.270
$\text{Na}_2\text{HPO}_4.\text{H}_2\text{O}$	Disodium hydrogen	0.170
	phosphate	
${\rm Na_3C_6H_5O_{7P}.2H_2O}$	Sodium citrate	0.060
	dihydrate	
H2NCH2CO2H	Glycine	0.450
$H_2SO_4$ s.g. 1.84	Sulphuric acid	0.050

The above materials were diluted to 1 litre with distilled water to form a physiological-like saline solution.

0.500 grams  $\pm$  0.0003 grams of chopped fibre was weighed into a plastic centrifuge tube and 25 cm<sup>3</sup> of the above saline solution added. The fibre and saline solution was shaken well and inserted into the shaking incubator water bath maintained at body temperature (37°C  $\pm$  1°C). The shaker speed was set at 20 cycles/minute.

After the desired period (usually 5 hours or 24 hours)

the centrifuge tube was removed and centrifuged at ≈ 4500 revs/minute for approximately 5 minutes. Supernatant liquid was then drawn off using a syringe and hypodermic needle. The needle was then removed from the syringe, air expelled from the syringe, and the liquid passed through a filter (0.45 micron cellulose nitrate membrane filter paper [WCN type from Whatman Labsales Limited]) into a clean plastic bottle. The liquid was then analysed by atomic absorption using a Thermo Jarrell Ash Smith - Hiefje II machine.

The operating conditions were:-

ELEMENT	WAVELENGTH(nm)	BAND WIDTH	CURRENT (MA)	FLAME (Nitrous Oxide + Acetylene)
Al	309.3	1.0	8	Fuel Rich
${\tt SiO}_2$	251.6	0.3	12	16 65
CaO	422.7	1.0	7	Fuel Lean
MgO	285.2	1.0	3	H H

The procedure and standards adopted for determining the above elements were as set out below.

 $SiO_2$  can be determined without dilution up to 250 ppm concentration (1 ppm = 1mg/Litre). Above this concentration an appropriate dilution was made volumetrically. A 0.1% KCl solution (0.1g in 100 cm<sup>3</sup>) was added to the final dilution to prevent ionic interference. NB If glass apparatus is used, prompt analysis is necessary.

From a stock solution of 1000 ppm pure ignited silica (99.999%) (fused with  $Na_2CO_3$  at 1200°C for 20 minutes in a platinum crucible (0.2500g  $SiO_2/2g$   $Na_2CO_3$ ) and dissolved in dilute hydrochloric acid (4 molar) made up to 250cm<sup>3</sup> with distilled water in a plastic volumetric flask) the following

standards were produced:-

STANDARD (PPM SiO <sub>2</sub> )	STOCK SOLUTION (cm <sup>3</sup> )
10.0	1.0
20.0	2.0
30.0	3.0
50.0	5.0
100.0	10.0
250.0	25.0

Add 0.1% KC1 to each standard before making to 100cm3.

Aluminium may be measured directly from the sample without dilution. Standards of 1.0, 5.0 and 10.0 ppm Al may be used. For calibration readings are multiplied by 1.8895 to convert from Al to  $Al_2O_3$ .

A standard Al atomic absorption solution (e.g. BDH 1000 ppm Al) was bought and diluted using an accurate pipette to the desired concentration. 0.1% KCl was added to prevent ionic interference.

Calcium may require dilutions on the sample before determination can be carried out (i.e.  $\times$  10 and  $\times$  20 dilutions). Dilutions must contain 0.1% KC1.

A standard Ca atomic absorption solution (e.g. BDH 1000 ppm Ca) was diluted with distilled water and an accurate pipette to give standards of 0.5, 4.0 and 10.0 ppm. 0.1% KC1 is added to prevent ionic interference. To convert readings obtained from Ca to CaO a factor of 1.4 was used.

Magnesium may require dilutions on the sample before determinations can be made (i.e.  $\times$  10 and  $\times$  20). Add 0.1% KC1 to each dilution. To convert Mg to MgO multiply by 1.658.

A standard Mg atomic absorption solution (e.g. BDH 1000 ppm Mg) was diluted with distilled water and an accurate

pipette to give standards of 0.5, 1.0 and 10.0 ppm Mg. 0.1% KCl was added to prevent ionic interference.

All stock solutions were stored in plastic bottles.

The results of the tests are indicated in Table 2.

TABLE 2

BODY FLUIDS SOLUBILITY

(ppm)

	<del></del>	SiO <sub>2</sub>	Ca	aO	1	lg0
	5 <b>h</b>	24h	5 <b>h</b>	24h	5h	24h
SW-A	98	120	63	56	33	66
SW-A1	83	141	32	70	21	70
SW-A2	130	202	43	73	100	177
SW-B1	58	77	10	38	5	9
SW-B2	64	121	27	55	5	10
SW-B3	138	192	80	46	8	21

Fibres with the best solubility (SW-A2 and SW-B3) were then tested, after annealing at varying temperatures, and compared with the comparative examples of Table 1. The results are shown in Table 3.

It can be seen that for the SW-A2 fibre, with increasing annealing temperature, the silica solubility drops progressively. In contrast the SW-B3 composition shows no loss in solubility up to 800°C and although a reduction in solubility is shown above that temperature it is not as dramatic as for SW-A2. Despite this difference in solubility it is to be noted that only the needled GF fibre shows a

\$15\$ comparable silica solubility and that material melts at 700  $^{\circ}\text{C}$  .

TABLE 3

<u>Fibre</u>	Condition	<u>S</u>	olubili	ty Anal	Lyses		
		CaO	(ppm)	MgO	(ppm)	Sio	(ppm)
			24hrs				24hrs
SW-A2	As received	58	37	37	3	8 <b>9</b>	130
SW-A2	600°C, 48hrs	33	56	27	43	60	108
SW-A2	800°C, 48hrs	35	53	17	30	43	87
SW-A2	1000°C, 48hrs	7	3	3	2	11	21
	As received		69	7	22	22	100
SW-B3	600°C, 48hrs			12	22	5 <b>5</b>	130
SW-B3	800°C, 48hrs	41	90	3	7	24	144
SW-B3	1000°C, 48hrs	18	40	3	3	17	60
CRBT	As received	10	8	6	3	5	3
CHBT	As received	16	10	7	3	4	0.3
Glass Fibr	ce As received	14	17	5	3	5	7
Needled G	F As received	17	34	8	15	6 <b>6</b>	85
Needled G	600°C, 48 hrs	11	26	7	10	19	37
Mineral Fi	ibre As received	16	16	7	6	8	9

[The Glass Fibre and Needled Glass Fibre had the compositions shown in Table 1.]

The user is primarily concerned with the solubility of the fibre as received as it is in this condition that most handling occurs; as received both SW-A2 and SW-B3 fibres have extremely high solubility. Even after exposure to 800°C and 1000°C these fibres have solubilities much higher than other high temperature use fibres.

To investigate the reasons underlying the difference in solubilities after high temperature annealing between the SW-A2 and SW-B3 fibres qualitative X-ray diffraction was done on the fibres. The results are indicated in Table 4 and it can be seen that the SW-B3 fibre forms pseudowollastonite and wollastonite, whereas the SW-A2 fibre forms diopside. It appears therefore that the crystalline diopside has a lower solubility in physiological saline solution than the crystalline pseudowollastonite and wollastonite material precipitated from the SW-B3 fibre.

TABLE 4

Sample	Condition	Oualitative XRD
SW-A2 SW-A2	600°C, 48 hours 800°C, 48 hours	Amorphous Amorphous with small amount Diopside
SW-A2	1000°C, 48 hours	Diopside
SW-B3	600°C, 48 hours	Amorphous
SW-B3	800°C, 48 hours	Amorphous
SW-B3	1000°C, 48 hours	Pseudowollastonite & Wollastonite

Various of the fibres were then tested for their shrinkage characteristics. Table 5 shows the results of Shrinkage tests on all the test fibres and on some of the comparative fibres. These results were obtained by proposed ISO standard ISO/TC33/SC2/N220 (equivalent to British Standard

BS 1920, part 6,1986) with some modifications to account for small sample size. The method in summary comprises the manufacture of vacuum cast preforms, using 75g of fibre in 500cm³ of 0.2% starch solution, into a 120 x 65mm tool. Platinum pins (approx 0.1-0.3mm diameter) were placed 100 x 45mm apart in the 4 corners. The longest lengths (L1 & L2) and the diagonals (L3 & L4) were measured to an accuracy of ±0.01mm using a travelling microscope attached to a steel rule with a vernier scale. The samples were placed in a furnace at temperature and left for 24 hours. The shrinkage values are given as an average of the 4 measurements.

TABLE 5

LINEAR SHRINKAGE (%)

(24h at temperature)

Temp.°C	SW-A	SW-Al	SW-A2	SW-B1	SW-B2	SW-B3
730	1.45		1.43	1.02	0.22	
870				0.41		
900			1.07			1.07
1000		1.04	1.3	0.51	0.6	1.1
1100		0.71	1.8		0.73	2.2
Maximum						
Service Temperatur °C	850 e	1050	1050	1050	1050	1000

It can be seen that in SW-A, SW-A1, SW-A2, SW-B1, SW-B2 and SW-B3, owing to the increase in molar volume on crystallisation, the linear shrinkage at the maximum service temperature is less than 3.5%.

Table 6 shows the results of a further series of shrinkage tests made in the same way.

TABLE 6

Sample	Measurement Direction c.f. Roll Direction	Test Temperatures °C	Linear Shrinkad Range	ges % Mean
			<u> </u>	
SW-A2	Parallel	850	1.1-1.4	1.2
SW-A2	Perpendicular	850	0.7-1.5	1.3
SW-A2	Parallel	900	0.5-1.1	0.9
SW-A2	Perpendicular	900	1.9-4.5	3.0
SW-A2	Parallel	1000	0.5-2.9	1.3
SW-A2	Perpendicular	1000	1.7-2.9	2.2
SW-A2	Parallel	1100	0.7-1.5	1.0
SW-A2	Perpendicular	1100	1.0-2.6	1.8
SW-B3	Parallel	900	1.6-1.8	1.7
SW-B3	Perpendicular	900	1.4-2.4	2.1
SW-B3	Parallel	1000	1.6-2.3	1.9
SW-B3	Perpendicular	1000	1.0-2.3	1.7
SW-B3	Parallel and	1100	Complete	Melting
	Perpendicular		(Lantern remnant)	

To ascertain the applicability of these tests to long term usage a series of cyclic shrinkage tests were undertaken on the materials and the heating schedule used for these cyclic

tests is shown in Figure 3.

The results of the tests are shown in Tables 7 & 8 (the two figures given for SW-B3 are due to slight differences in chemical analysis [the fibre at the end of a production run of fibre tends to have slightly differing composition to that at the beginning of a production run of fibre]).

As a further comparison with the above discussed materials a melt was made comprising 55% SiO<sub>2</sub>, 29.9% CaO and 18.6% MgO. Fibres made using this composition had a maximum service temperature of 700°C and melted at 800°C.

As these results were encouraging the applicants conducted a further and extensive series of tests, concentrating on the SW-A2 and SW-B3 compositions, to ascertain the reproducibility of these results and the boundaries of the useful compositions.

Table 9 (three pages) below gives the compositions of a series of melts, ranked on silica content, and showing the shrinkage figure after exposure to 1000°C for 24 hours (1st column) and 800°C for 24 hours (2nd column). shrinkages were measured by the same method as the shrinkages given above but measurements were made with a travelling microscope with a digital linear scale accurate to  $\pm$  5 $\mu$ m. can clearly be seen that all fibres with a silica content of less than 58% have a shrinkage at 1000°C of greater than 3.5% save two (B3-3 and 708). These fibres, together with some fibres with a silica content of greater than 58% although showing a reasonable figure at 1000°C, show a very poor figure at 800°C. Compositions with an SiO2 content of greater than 70% appear to fiberise poorly. This may be because such compositions have two liquids in the melt as appreciated from Fig.1

TABLE 7

CYCLIC SHRINKAGE (LINEAR)

(왕)

Product	1000°C	1100°C	24h at 1000°C
No. cycles	58	42	
CRBT	.2.0	2.7	1.9
CWBT	15.0	13.3	12.1
SW-A2	0.33	2.0	1.3
SW-B3	1.00	1.67	1.1
SW-B3	0.33	0.67	1.1

Accuracy: + or - 0.33%

TABLE 8

CYCLIC SHRINKAGE

(%)

-		LINEAR SHRINKAGE		THICK SHRIN	
PRODUCT	1000°C	1100°C	24 h at 1000°C	1000°C	1100°C
No. cycles	104	100		104	100
CRBT CWBT SW-A2 SW-B3 SW-B3	1.47 14.4 1.5 1.73 1.47	3.1 15.2 2.1 1.63 1.77	1.9 12.1 1.3 1.1	0.47 38.63 8.58 7.24 7.02	11.19 32.14 8.75 7.57 7.16

Accuracy: + or - 0.3%(%)

## PARI.R 9

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itions	ζ0% F	0.05	0.05	0.05	60.	0.05	0.05	0.05	0.05	.07	0.05	0.05	90.	.05	.10	.05	.08	.05	.05	.05	.11	.07	10	.05	80.	.05	80.	.05	.05	0.12	.05
d Compositi ight %)	Na 20% K	.05	0.	.05	۳.	.05	۲.	.05	80.	7	•	111		7	.3	0.	<del>د</del> .	Ţ,	4	0.	4.		4.	0.		0	4.	7	٦.	0.28	۲.
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Analy	MgO% 1	7.1	•	9.6	•	2.3	۳.	2.3	9.6	7.0	٠ د	2.8	3.7	2.7	.7	9.5	7.0	2.6	.5	1.7	0.4	9.	8	0.	9.3	8	1.7	T.	.7	4.88	.2
	ca0%	0.	4.		4.	2.6	۳.	4.		8.74	.58	4.61	3.27	5.17	66.	8.12	0.98	7.45	.58	8.45	7.76	4.9	9.9	4.87	1.28	6.22	9.82	5.6	8.74	25.81	6.86
	$\mathrm{Sio}_2 \$$	8.0	3.4	3.0	2.3	2.2	1.4	1.2	1.1	0.8	0.4	0.0	9.4	9.2	8.7	8.6	8.5	8.3	8.1	7.6	7.5	7.5	7.2	9.9	6.1	6.1	5.8	5.7	5.6	65.50	5.3
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TABLE 9 (continued)

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Shrinka	1000	0	1.0		3.4	3.2	9.0	7.3	1.0	7.4	2.7	1.9	1.0	21.6	0	1.3	6.1	0.8	1.4	9.0	6.0	1.8	3,3	1.3	0.3	7.5	1.8	2.5	1.8	1.6	18.9
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	${\rm Sio}_2 {\rm \$}$	5.2	5.1	5.0	4.8	4.1	4.1	4.1	4.0	3.7	3.6	3.6	3.5	3.2	2.6	2.6	2.3	۳. م	6.	1.8	1.7	1.6	. 3	1.3	1.3	8.0	7.0	0.3	e. o	60.28	2.5
	Melt	_	ריי	21	2-3	CA .	3-3	2-3	3-2	3-5	2-3	2-9	2	3-6	23	57	2	3-1	2	່ຽນ	3-1	~	2-2	$\sim$	24	3-2	2-1	3	2-1	B3-9	2-

TABLE 9 (continued)

			Anal	Analysed Co (Weight	mpos	itions			Shrinka at °C	ıkage °c	Rationa Composi	l i t i	sed
Melt	$\sin_2 k$	Ca0\$	Mg0&	A12038	Na208	K20\$	Fe2038	Zro2%	1000	800	$\sin_2 x$	Ca0%	MgO%
Ċ	0.1	2.5	8	.7	3	.1	.2	8			2	<u>س</u>	
932	59.85	21.60	15.65	1.50	90.0	<0.0>	0.18	0.11	12.8		61.6	22.3	16.1
92	9.8	4.3	4.	۲.	۲.	0.	•		•		0	4	2
3-3	9.5	1.6	9.	Q.	۳.	Ċ.	~	4.	•	15.3			
e	9.5	8.1		4.	Э	•	۲.	•	•		<del>,</del>	6	•
3-2	4.6	0.9	9	9	4.	<u>.</u>	7		•		1:	7.	•
<del></del>	9.0	2.0	7	0.	?	•	4.	•	•	•	6	8	•
96	8.8	0.9	۲.	7	4	•	0.	7	•	1.4	1.	2	•
A2-19	8.7	8.4		ა.	?	0.		<0.05	•		9.09	19.0	20.4
œ	8.6	5.0	6	٦.	4	•	7	•	•	•	0	ហ	•
9	8.3	3.1	0.	7.	7	0	-	٦.	•	1.9	9.	щ •	•
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c	7.5	2.7	0.	9	.5	۲.		•	0		о О	щ •	•
12	7.5	5.3	•	•	4	0.	۲.	•		3.4	60	•	•
$\sim$	7.3	9.9	4.	. 7	ı.	0.11	· J	0	0		9	8	•
3-2	6.9	0.5	4.	•	4.	۲.	•	0.	6	26.6	8	1:	•
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734	6.5	•	•	•	0	0.	4	0.	•	-	8	•	•
7	6.1	4.4	6.8	5	0	<0.05		0.	4.		7	5.	7
C	5.9	6.0	4	Ŝ	4.	•	<b>.</b>	4.		34.7	8	7.	•
0	5.2	2.7		•	.3	<0.05	7	<0.05	ij	•	<u>ي</u>		0
71	4.6	4.0	9	3	<b>—</b>	<0.05	7	<0.0>	•		io.	4.	•
									-	*			

\*Poor fibres containing a lot of shot. All other constituents < 0.1%. Too poor to test for solubility or shrinkage

# TARIE 10

			Anal	ysed (Weig	Composit pt %)	itions			Solubilit ppm	ilit pm	ties	Ratic Compo	ionalis	sed
Melt	$\sin_2 $	Ca0%	Mg0%	Al2038	Na20%	K20%	Fe2038	$2ro_2$ %	$\sin_2$	CaO	Мдо	$\sin_2 $	Ca0%	Mg0%
2-2	3.0	2.0	7.1	٦.	0.	0.0		.7	<b>F</b>			0	8	7
2-1	3.4	12.4	0.0	7	0.0	0.	۲.	.2	113	45	55	9	•	•
2-3	3.0	6.3	9.6	7	0.	0.0	• 1	<del>د</del> .	-			3	•	9.
3-3	2.3	23.4	9.0	۳.	٤,	0.		. 7				5.	٠	•
2-1	2.2	12.6	2.3	7	•	0.	7	.2	0			4	3	2.
1	1.4	9.3	. 3	۳.	۲.	0.	?	8		36	75	<del>ب</del>	•	•
2-2	1.2	4.4	2.3	4	0.	0.		.2				5	•	2.
2-2	1.1	6.7	9.6	.2	0.	•	<u>.</u>	•	0			2	•	0
3-2	9.8	18.7	7.0	4.	.2	0.	<u>.</u>	.7				ж •	•	•
2-1	<b>7.4</b>	11.5	ა.		0.	0.	<u>.</u>	3				<del>ن</del>	2	•
2-3	0.0	4.6	2.8			0.		0.				٠.	4.	ж :
3-1	9.4	23.5	3.7	4.	۳,	0.	-	4.	4			٠;	4.	<del>ن</del>
9-7	9.2	15.1	. 7	•	.2	0.	<u>.</u>	۲.	2			1.	•	٠
3-1	3.7	24.9	1.7	9.	٤,	4	?	0.	3			2	9	;
2-2	9.6	8.1	7.	.2	0.	0		6.	8			7	ж •	٠
_	8.5	20.9	7.0	.7	.3	0.	۲.	. 7	2			;	;	7
59	8.3	17.4	2.6	0.	۲.	0	ب	0	_			6	7	5
2-2	8.1	11.5	6.5	4.		0.	?	9	9			0	5	٠
~	67.62		21.72	0.32	0.09	<0.05	0.15	0.64	103	56	72	69.2	8.6	22.2
3-3	7.5	27.7	0.4	4.	4.	-		•	~			0	თ	٠
3-1	7.5	24.9	•	4.	<b>.</b>	0		9	9			0	ហ	٠
3-1	7.2	26.6	ω.	. 7	4.		?	ທ	9			0	7.	;
2-1	9.9	14.8	6.0	۲.	0.	•	۲.	o.	S				3	•
2-1	6.1	16.2	α.	4.	0.	0	0	0	2			7.	9	•
3-2	6.1	21.2	9.3	R.		0.	;	.5	7			œ	ċ	•
3-1	5.8	29.8		4.	4.	0.	7	3	Ω			7	0	•
19	5.7	25.6	7	0.	.2	0.	.2	0	$\boldsymbol{\vdash}$			9	5	8
2-	5.6	18.7			.1	0.	۲.		2			9	ق	•
3-	5	25.8	4.8	0	.2	<u>.</u>	۲.	7.						
A2-8	5.3	16.8	7	7		0.	<b>:</b>	۲.				67.7	17.5	14.8
											7			

TABLE 10 (continued)

Berl Sio_2* Cao* Myo* Algo* Na_2o* Na_2o* Fe_2o_3* Zro_2* Sio_2 cao Myo Sio_2* Cao* Myo Sio_2* Cao* Myo Myo Sio_2* Cao* Myo Myo Sio_2* Cao* Myo Myo Si				Anal	Analysed Compositions (Weight %)	mposit %)	ions	•		Solubilities ppm	bilit ppm	ies	Rationa Composit	17.17	sed
Part	elt sio	& Ca	9/0	0	12038	~	20	e <sub>2</sub> 0 <sub>3</sub>	$ro_2$	io	a	Мдо	$io_2$	aO	Mg0%
9-14         65.11         24.91         5.54         0.58         0.43         0.09         0.19         0.61         153         68         27         68.08         20.08         0.09	18 65.	23 27	14	6	0.0	.2	0.0	.2	. 4	0	47	20	5.	7.	
65 08         27 26         5.31         0.06         0.17 <0.05         0.08         0.00         153         65 08         20         65 0.3         27.26         5.73         0.06         20.06         0.00         0.11         0.08         0.13         11         66.4         26.6         27.9         5.2           2-34         64.18         6.6.3         26.20         0.23         0.064         0.045         0.014         0.28         16.3         18         16.6         5.2         1.01         0.08         0.11         160         28         56.6         1.01         0.08         0.13         0.04         0.13         0.04         0.08         10.1         160         18         10         18         1         160         8.2         1.01         10         10         0.09         0.13         0.04         0.04         0.13         0.01         0.09         0.01         0.01         0.05         0.01         0.05         0.01         0.05         0.01         0.01         0.05         0.05         0.05         0.05         0.05         0.05         0.05         0.05         0.05         0.05         0.05         0.05         0.05         0.05         0.05	3-14 65.	11 24	91	S.	ល	4.	0.	. 1	9.	5	67	27	8	6.	•
2-34         64.85         6.63         26.20         0.23         0.06         <0.05         0.11         0.80         39         11         119         66.4         6.8         26.20           2-21         64.16         13.74         19.98         0.34         0.17         <0.05	21 65.	08 27	26	۳.	0.	7	0.0	•	0.0	2	68	20	9	7.	•
$\begin{array}{cccccccccccccccccccccccccccccccccccc$	2-34 64.	85 6	63 2	.2	7	•	0.0	<del>-</del>	8	39	11	7	9	6.	9
1-30         64.13         31.93         0.37         0.64         0.45         0.09         0.14         0.28         163         83         3         66.5         33.1         0.75           2-35         64.12         8.88         24.88         0.29         0.10         0.016         0.14         0.47         61         18         16         65.5         9.1         25.0           3-56         63.44         25.41         4.68         3.97         0.26         0.12         0.15         0.05         102         33         24.0         9.1         20.0         20.1         0.15         0.05         102         33         52         65.0         16.4         18         16         66         64.9         24.0         9.1         20.0         0.10         0.11         0.05         10.0         0.11         0.05         10.0         0.11         0.05<	2-21 64.	16 13	74 1	ç.	۳,		0.0	٦.	. 1	9	28	56	5.	4.	0
$ \begin{array}{cccccccccccccccccccccccccccccccccccc$	3-30 64.	13 31	93	.3	9.	4.	0.		3	9	83	e	9	<del>ب</del>	0
3-54         64.09         23.26         9.33         0.56         0.36         0.09         0.16         0.30         101         44         31         66.3         24.0         9.33           3-5A         63.74         25.41         4.68         3.97         0.26         0.12         0.17         0.058         48         43         11         6.06         18.20         1.0         1.0         1.0         0.11         <0.05	2-35 64.	12 8	88 2	ω.	7.		0.	۲.	4.	9	18	0	5	•	5.
3-5A         63.74         25.41         4.68         3.97         0.26         0.12         0.17         0.58         48         43         11         4.68         3.97         0.26         0.11         0.07         0.15         <0.05         102         33         52         65.0         16.4         18.1           2-3         63.66         16.06         18.21         0.40         0.11         0.07         0.11         <0.05	3-23 64.	09 23	56	٠,	3	۳.	0.	7	.3	0	44	31	9	4.	•
$ \begin{array}{cccccccccccccccccccccccccccccccccccc$	3-5A 63.	74 25	41	9.	è	7.	4	τ.	5	4	43	11			
$\begin{array}{cccccccccccccccccccccccccccccccccccc$	2-30 63.	68 16	06 1	7	4.		0.	۲.	•	0	33	52	5	6.	 ∞
$\begin{array}{cccccccccccccccccccccccccccccccccccc$	2-9 63.	66 21	44 1	9.	4.	<b>.</b>	۲.	~	0.	9	67	99	4.	;	$\sim$
$\begin{array}{cccccccccccccccccccccccccccccccccccc$	2-12 63.	56 16	55 1	0.	<del>د</del> .	٠,	0.	0.	۰.	9	47	99	4.	9	8
23         62.61         29.79         5.44         <0.05         0.17         <0.05         0.13         <0.05         141         62         17         64.0         30.4         5.7           2-25         62.60         20.92         15.22         0.20         <0.05	3-6A 63.	24 24	83	٠.	. 7	.2	۲.	۲.	۲.	2	7	26			
$\begin{array}{cccccccccccccccccccccccccccccccccccc$	23 62.	61 29	79	4.	0.		0.0	٦.	0.0	4	62	17	4.	0	٠
$\begin{array}{cccccccccccccccccccccccccccccccccccc$	57 62.	60 20	92 1	7	7	0	0.0	۲.	0.0	œ	62	7	<del>ب</del>	+	ເດ
$\begin{array}{cccccccccccccccccccccccccccccccccccc$	2-25 62.	36 10	99   2	۲.	۳.		0	~	0	9	35	0	ст С	<del>.</del>	4.
25-7         61.98         23.37         11.98         0.44         0.25         <0.05         0.10         0.23         178         59         63         63.7         24.0         12.           25         61.83         28.13         7.54         0.10         0.36         <0.05	3-13 62.	33 30	62	•	ស	•	0.	_	9	S	41	7	ე.	5	5
25         61.83         28.13         7.54         0.10         0.36         <0.05         0.17         <0.05         186         76         35         63.4         28.9         7.53           33-11         61.71         33.25         2.33         0.69         0.52         0.12         0.25         162         28         15         63.4         28.9         7.2           3-24         61.62         25.53         9.73         0.58         0.10         0.17         <0.05	2-7 61.	98 23	37 1	9	4.	.2	0.0		7	7	59	63		4.	5
3-1161.7133.252.330.690.520.120.250.59162281563.434.22.3-2461.6225.539.730.580.380.100.17<0.05	25 61.	83 28	13	r.	4	.3	0.0		0.	8	26	35	<del>.</del>	<del>α</del>	•
3-2461.6225.539.730.580.380.100.17<0.05	3-11 61.	71 33	25	۳.	•	.5	4	7	ທ	9	28	15		4.	~•
2-24 61.38 13.62 22.74 0.31 0.16 <0.05 0.14 0.08 136 27 87 62.8 13.9 23.  22 61.33 31.08 5.25 0.05 0.20 <0.05 0.16 <0.05 185 81 23 62.8 31.8 5.  24 61.32 19.78 14.54 2.57 <0.05 <0.05 0.09 0.66 77 36 37 64.1 20.7 15.  2-14 60.74 25.30 1.66 0.28 0.18 0.05 0.13 0.15 127 57 43 62.2 25.9 11.  2-14 60.32 32.27 3.99 1.74 0.32 0.10 0.15 0.09 119 72 16  2-11 60.32 24.28 13.24 0.25 0.18 <0.05 0.09 0.08 149 54 51 61.7 24.8 13.24 0.25 0.18 <0.05 0.09 0.08 149 54 51 61.7 24.8 13.24 2.50 0.76 0.50 0.13 0.28 0.49 175 8 5 62.0 35.4 2.22 2.23 60.20 18.59 18.78 0.48 0.19 0.08 0.15 0.05 192 35 47 61.7 19.1 19.	3-24 61.	62 25	53		.5	۳.	4		•	œ	37	57	ن	9	
22     61.33     31.08     5.25     0.05     0.20     <0.05     0.16     <0.05     185     81     23     62.8     31.8     5.       24     61.32     19.78     14.54     2.57     <0.05	2-24 61.	38 13	62 2	.7	<del>.</del>	4	0.0		•	C	27	87	ς.	<del>ن</del>	<del>ن</del>
24     61.32     19.78     14.54     2.57     <0.05     <0.05     0.09     0.66     77     36     37     64.1     20.7     15.7       3-2A     60.83     32.30     0.48     4.15     0.35     0.15     0.19     0.12     58     61     5     64.1     20.7     15.       2-14     60.74     25.30     11.66     0.28     0.18     0.05     0.13     0.16     127     57     43     62.2     25.9     11.       3-1A     60.32     32.27     3.99     1.74     0.32     0.10     0.15     0.09     119     72     16       2-11     60.32     24.28     13.24     0.25     0.18     <0.05	22 61.	33 31	08	7	0.	?	0.0	۲.	•	8	81	23	7	;	ъ.
3-2A60.8332.300.484.150.350.150.190.12586154362.225.911.2-1460.7425.3011.660.280.180.050.130.150.16127574362.225.911.3-1A60.3232.273.991.740.320.100.150.090.0911972162-1160.3224.2813.240.250.18<0.050.130.280.49149545161.724.813.3-960.2834.492.500.760.500.130.280.491758562.035.42.2-2360.2018.5918.780.180.150.050.150.0519.119.119.1	24 61.	32 19	78 1	<u>د</u>	3	0.0	0.0	0	9.	7	36	37	4	0	کا
2-14 60.74 25.30 11.66 0.28 0.18 0.05 0.13 0.16 127 57 43 62.2 25.9 11. 3-1A 60.32 32.27 3.99 1.74 0.32 0.10 0.15 0.09 119 72 16 2-11 60.32 24.28 13.24 0.25 0.18 <0.05 0.09 0.08 149 54 51 61.7 24.8 13.2 3-9 60.28 34.49 2.50 0.76 0.50 0.13 0.28 0.49 175 8 5 62.0 35.4 2. 2-23 60.20 18.59 18.78 0.48 0.19 0.08 0.15 0.05 192 35 47 61.7 19.1 19.	3-2A 60.	83 32	30	4.		.3	۲.	۲.	۲.	ນ	61	S			
$ \begin{array}{c ccccccccccccccccccccccccccccccccccc$	2-14 60.	74 25	30 1	9	7	Ţ.	0.		٦.	3	22	43	5	5.	7
2-11     60.32     24.28     13.24     0.25     0.18     <0.05	3-1A 60.	32 32	27	6.	.7	<b>.</b>	۲.		0	H	72	16			
3-9 $60.28$ $34.49$ $2.50$ $0.76$ $0.50$ $0.13$ $0.28$ $0.49$ $175$ $8$ $5$ $62.0$ $35.4$ $2.$ $2-23$ $60.20$ $18.59$ $18.78$ $0.48$ $0.19$ $0.08$ $0.15$ $0.05$ $192$ $35$ $47$ $61.7$ $19.1$ $19.$	2-11 60.	32 24	28 1	2	.2	٠,	0.	0.	0	4	54	51	;	4.	٠ ش
2-23   60.20   18.59   18.78   0.48   0.19   0.08   0.15   0.05   192   35   47   61.7   19.1   19	3-9 60.	28 34	49	5	.7	3		.2	4.	6			<i> &lt; &lt;</i>	س	5
	2-23 60.	20 18	59 1	.7	4.	۲.	0.	٠,	0	9			<del>.</del>	9	9

TABLE 10 (continued)

sed	Mg0%		16.1	5		•	•	•	•	•	•	6.2	•	•		•		•	ж •	•	7.	7		0	. 1
	ca0%	3	22.3	4.		6	7	3	2	•	5	33.9	4.	δ.	ъ.	6.	α,	j.	7.	•	4.	5	7.	<del>с</del>	4.
Rational Composit	$\sin_2 %$	2	61.6	0		<del></del>	7	9.	1.	0	0	59.9	e.	9	9	6	9	8	6	7	8	7	٠ ش	کا	
ies	MgO					33		30	37	99	19	27	169	23	40	22	16	41	57	73	99	47	18		62
bilit ppm	CaO	84	41	67	25	48	29	73	48	42	54	99	16	69	29	24	25	64	47	55	52	42	14	31	55
Solubilit ppm	$\sin_2$	146	92	140	C	133	0	4	9	$\sim$	$\infty$	3	0	S	œ	9	6	4	9	4	3	0	æ	9	133
	$ZrO_2$	8	٦.	4	4.	<0.05	·	•	?	•	0.	4	0.	0.	0.	0.	•	•	0.	0.05	0.	0.	4.	0.	0
	Fe2038	. 2	:	0	.2	0.14	?	4.	0.				.2		.2			•	7	۲.		۲.	۳.	۲.	.2
itions	K20%	. 1	0	0	4	0.08	4	0.	0	0.	0.	0.	0.	0.	٠,	0.	۲.		۲.	0	0.	0	0	0.	0.
npos &)	$Na_20$ %	5	0.	۲.	٠.	0.39	4.	3	۲.	3	~	۲,	•	7	ις.	7	٠.	4.	5	Ţ.	0	0.	4.	<b>.</b>	7
Analysed Cor (Weight	Al 2038	.7	.5	۲.	9	0.48	6.	0.	7.	.5		.2			o.	0.	. 7	9	•	.7	•	3	٠ د	0.	. 2
Ana]	Mg0%	æ	•	5.4	•	~	•	?	4	. 7	Q.	•	0.	æ	0.	•	4.	4.	3.0	٤.	7.0	6.8	4.4	.7	19.66
	ca0%	2.5	1.6	4.3	1.6	28.15	6.0	2.0	0.9	8.4	5.0	3.1	3.9	4.6	2.7	5,3	9.9	0.5	6.2	ы О	3.6	4.4	6.0	2.7	4.0
	${\rm sio}_2 \$$	0.1	9.8	9.8	9.5	59.53	4.6	9.0	8.8	8.7	8.6	8.3	7.7	7.7	7.5	7.5	7.3	6.9	6.9	6.8	6.5	6.1	5.9	5.2	4.6
	Melt	(	932	9	3-3	B3-25	3 - 2	$\boldsymbol{\vdash}$	9	2	8	694	9	9	$\mathbf{c}$	$\leftarrow$	3 - 2	C	2-1	971	C	7	C	0	7

There are several anomalies, namely compositions B3-6A, A2-25, A2-24, A2-23, B3-2A, B3-3A, A2-19, and 932. All of these have an  $SiO_2$  content of > 58% but a high shrinkage.

On the assumption that the minimum silica level for satisfactory shrinkage varies with MgO content the applicants have determined that fibres with a silica content (in weight percent) that fail to meet the following expression do not have satisfactory shrinkages at either or both 800°C and 1000°C:-

$$SiO_2$$
 >58% - (for MgO =< 10%) and  $SiO_2$  >58% + 0.5(%MgO -10) - (for MgO >= 10%)

The applicants have further found that the  ${\rm Al}_2{\rm O}_3$  content is important. From their studies it appears that the maximum  ${\rm Al}_2{\rm O}_3$  content lies somewhere between 2.57% and 3.97%. The applicants have found that with increasing alumina levels the first material to crystallise is calcium aluminate and this possibly forms a liquid phase that assists flow and hence shrinkage.

Table 10 shows, for the same compositions as Table 9, 24 hour solubilities for each major constituent. It can be seen that all of the compositions have high solubilities.

As mentioned above use of CaO in forming calcium containing fibres is inconvenient and can be hazardous. The applicants investigated use of mixed oxide materials that would avoid the handling of CaO. A fibre was made by admixture of magnesia with silica and wollastonite (CaSiO3).

The raw materials used to make the melt comprised:-

Pennine Darlington Heavy Magnesia (#200)

MgO	-	92.60%
CaO	-	1.75%
$Fe_2O_3$	-	0.04%
sio <sub>2</sub>	-	0.20%
Cl	-	0.25%
so <sub>3</sub>	-	0.70%
LOI	-	4.50%

Partek's Finnish Low Iron Wollastonite (#200) (U.K. agent - Cornelius Chemical Co., Romford, Essex)

$sio_2$	-	51.80%
CaO	-	44.50%
MgO	-	0.80%
Al <sub>2</sub> 0 <sub>3</sub>	-	0.60%
$Fe_2O_3$	-	0.30%
Na <sub>2</sub> O	-	0.10%
к <sub>2</sub> 0	-	0.05%
TiO <sub>2</sub>		0.05%
S	-	0.02%
MnO	-	0.01%
P	-	0.01%
F	-	0.01%
LOI	-	1.70%

Hepworth Mineral's Redhill T washed silica sand

SiO<sub>2</sub> - 99.0% min.

These constituents were mixed as 78.65% Wollastonite;  $19.25\% \, SiO_2$ ; and 3.6% MgO. This gave 0.4 - 0.5% of the final melt as  $Al_2O_3$ .

It was surprisingly found that in producing a melt using these constituents the current requirements were only two-thirds that for the use of the raw oxides.

Fibre was produced by blowing (although spinning and other methods can be used). 2 runs were performed with different blowing conditions.

Chemical analysis was undertaken by the Analytical Department at the applicant's subsidiary Morgan Materials Technology (hereinafter called  ${\tt M}^2{\tt T}$ ) using wet chemical techniques. Fibre diameters were measured using  ${\tt M}^2{\tt T}$ 's Galai particle analyser, with shape analysis software. Typically 40,000 fibres were analysed for each run.

The first result of note was the speed of melt reaction when using wollastonite as compared with lime. Also the current was seen to be very stable throughout the growth of the melt. If the current was lost whilst pulling the electrodes apart the current could be restored simply by pushing them back together again. This was not possible with the runs using lime.

#### Chemical Analysis

	<u>Others</u>	<u>Ca0</u>	MgO	Al <sub>2</sub> 0 <sub>3</sub>	SiO <sub>2</sub>	zro <sub>2</sub>	Na <sub>2</sub> O	<u>Total</u>
Run1	0.7	32.6	3.8	0.8	60.1	0.8	0.5	99.3
Run2	0.7	32.5	3.8	0.8	60.1	0.8	0.6	99.3

Runs 1 and 2 indicate the respective x-ray fluorescence analyses for each run.

#### Shrinkage Results (1000°C for 24 hours)

		<u>L1</u>	L2	<u>L3</u>	<u>L4</u>	<u>Av.</u>	Std.Dev.
Run	1	0.9	0.2	0.4	0.6	0.5	0.3
Run	2(A)	1.0	-0.2	0.7	0.6	0.5	0.5
Run	2(B)	0.5	0.2	0.0	0.4	0.2	0.2

### Solubility Results (ppm)

	<u>CaO</u>	<u>MgO</u>	<u>sio</u> 2
Run 1(5 hr)	67	10	95
Run 1(24hr)	84	17	146
Run 2(5hr)	39	7	72
Run 2(24hr)	73	17	186

### Fibre diameters

	<u>Mean</u>	Median	<u>100%&lt;</u>	%>5µm	%<1µm
Run 1	5.1 $\mu$ m	$3.4 \mu \mathrm{m}$	- 30μm	33%	13%
Run 2	$4.1 \mu \mathrm{m}$	$2.7 \mu m$	25µm	25%	19%

Accordingly it appears to be the case that by using what are cheaper ingredients than the pure oxides one can obtain a fibre that has as high a performance as using purer oxides and at much improved energy costs and safety. It is to be noted that this feature of the invention is not limited to saline soluble fibres and any oxide fibre that contains both calcium and silicon can advantageously be made with a calcium silicate, wollastonite being merely an example of such a silicate.

The previous description is directed towards high temperature usage of particular saline soluble fibres. The following is directed towards the prediction and use of saline soluble fibres. A series of fritted glass melts were made of the compositions shown in Tables 11A and 11B and quenched in water. Solubilities of the various components of the quenched melt were measured by the previously described method of atomic absorption. The solubilities were normalised to a specific surface area of 0.25m²/gram of fibre.

The free energy of hydration was calculated by normalising the chemical analysis to 100 weight %; making the assumption that simple silicates (MSiO $_3/M_2$ SiO $_3$ ) are present

and calculating the free energy of hydration contribution of each species; and summing to get the total free energy of hydration. The data in Tables 11A & 11B is also presented in Fig.4. It can be seen that the fibres lie on a generally straight line bar four groups of materials that will be explained below.

Table 11 shows in each column the following:-

Fibre reference Composition Molar ratios

Moles/100 grams of melt

Species assumed (see below)

Calculated free energy of hydration of assumed species (kcal/100 grams)(see below)

Calculated free energy of hydration of assumed species (kJ/kg) (see below)

Solubility data (# indicates not measured [see below])

Specific surface area

Normalised solubility data

log normalised solubility

The base data on which calculation of the free energy of hydration was done is set out in Tables 12 which indicates free energies of hydration taken from the literature in the units kcal/mol and kJ/mol.

lable of free Energy of Hydration Values for Silicate Helts

### TABLE 11

	_	1000	_	in the	Hydration (kept/100ct)	Workers in the 1/101	177		_		_
CaSiO3	Ť	Ť	15.7	-	L'L'	0 701	200		5/2	Solverilley	"
A1203	A1203		2,63		? 6	)	1201	; <	D. 50	<b>;</b>	.63
CaO	CaO	_	10.5		-5.3	-219.7	Sio2	. ~			
					-11.6	.492.1	Total	2			
Casio3	Casio3		4.79		.7.7	-322.8	CaO	28	0.39	3	1.82
A1203	A1203		2.35		0.7	30.6	A1203	_			
0.363 Cao 3.63	Ge 3		3.63		9.4-	- 198.9	2018	77			
					-11.8	-491.1	lotal	105			
0.456 Casio3 4.56	CaSi03	_	4.56	i	-7.3	-307.3	CaO	55	0.36	16	1.08
0.237 A1203 2.37	A1203		2.37		0.7	30.8	A1203	9			
0.407 CaO 4.07	CaO		4.07		-5.3	-223.0	2015	\$\$			
					-11.9	5.667-	Total	110			
0.707 Casio3 7.07	CaSi03	<u> </u>	7.07		-11.4	-476.5	CaO	23	07.0	69	- 8
0.083 C.i.O 0.83	C''O		0.83			-45.5	1102	0			
0.165 1.02 1.65	1,02		1.65		2.6	110.4	s io2	67			
					6.6-	411.6	fotal	Ξ			
0.592 Srsio3 5.92	SrSi03		5.92		-14.4	7.709-	. 01S	~	0.37	549	2.40
0.030 \$10 0.30	015		0.30		8.9	-21.1	2005	367			
-					6.41 -	. 929.1	Total	369			
0.836 Casio3 8.36	Casio3		8.36		13.5	-563.5	CaO	1,	0.45	163	2.21
0.048 sio2 0.48	\$ 102		0.48		0.3	11.2	\$102	355			
					-15.2	-552.5	Lectur	294		-	
Ma2SiO3	Ma2SiO3		3.05		8.8	-367.5	H.12G	•	0.56	,	09.0
A1 203	A1203		3.90		1.2	20.7	A1203	4			
0.382 5:02 3.62	\$ 102	_	3.85		2.1	7.68	2015	~			
					-5.5	-227.4	local	۰			
0.373 Casio3 3.73	CaSi03		3.73	ì	0.9-	-251.4	CaO	=	0.62	8	0.00
A1203	A1 203		3.45		=	6.44	A1203	<b>4</b>			
0.358 si02 7.31	S io2		7.31		2.0	171.1	2015	~			
				1	-2.9	-35.4	fotal	2			
0.256 HgSiO3 2.56	HgSio3	_	2.56		-3.6	-149.0	M <sub>90</sub>		0.52	۰	0.78
A1 203	A1 203	_	3.45		1:1	44.5	A1203	~			_
0.656   sio2   6.56	S i 02		6.56		3.7	153.5	Si02	<b>-</b>			_
					1.2	0.67	Total	2			:
0.294 Srsio3 2.94	Srsio	-	76.2		-7.2	-300.2	Sr0	۰	0.50	9	0,.0
A1203	A1203	_	3.12		1.0	9.05	A1203	~			
S 102	S 102		3.33		1.9	17.9	2015	~			
		_			-4.3	-181.7	lotal	9			
0.330 Srsio3 3.39	SrSi03	_	3	٥	.0.3	-346.1	Sro	22	0.41	280	5.45
Casio3	Casio3		3.7	9	.6.1	-253.4	CúO	2		_	
2015		_			-	144 144	2102	155	_		_
	2015	_	•		-				_		

lable of free tnergy of Hydration Values for Silicale Melts

# TABLE 11 Continued

-	16 m	3	_							_			· ~	_	-			~				<u>.</u>	7	3							<	<u> </u>	•		<u>~</u>			
		50(141)						1.9B					2.35	_				1.75			-	1.38			_	- 1	-				70	:			1.15	_		
	MOI BLAIN SEC	Solubilley	28	,				ş					212					3				7.7				011	:				92				16			
3 3		1	o o				,	0.39				1	0.39				;	0.58				. s				0.48	?				0,40				0.36			
201			7	·	^ •	n (	8	2	2	-	22	871	147	2	3	123	33	2	_	<b>40</b> j	≥!: 	=	` ,	۰.	9 2	2	23	•	118	228	107	-	35	147	-	•	-	=
Solimilier	1	200	3 (	200	6103	7016	100	2 :	M <sub>1</sub> 0	A1203	2015	LOCAL	o vs	00 H	A1203	2015	TO COL	Sro	A1205	2015	10141	7	Hg0	A1203	2018	CaO	M90	A1203	2015	lutal	Sro	A1205	\$102	lotal	K20	MgO	A1203	2015
free Energy of	Hydration (* 174.0)	, cor.		7:5		2.161	7 007		23.65	7.6	۸. ۲. ۲. ۲. ۲. ۲. ۲. ۲. ۲. ۲. ۲. ۲. ۲. ۲.	5.525.	1.526.	9.0g1.	<b>5.</b>	61.8	445.0	\$.60\$.	1.03	1.501.	* 00.	5.7%1	9.6	4.55	9.71	- 260.8	-226.4	1.0	9.0	-437.2	-421.7	52.9	-101.4	-500.2	-305.4	-188.0	6.22	204.8
free Energy of	Hydration (kcal/100g)	.7.2	20-	0.7	4.3	0.11.	8.0-		9.0	7 6	3.5	6.21.	8.	? .	9.0	c · ·	9.01.	9.4	0.0	7.17	2.7.		)	6.4.		-6.2	-5.4	0.0	1.4	-10.2	1.01.	0.5	-2.4	-12.0	.7.3	-2.1	0.5	2.1
Moles	in 1kg	2.96	0.30	2.39	2.50		10.4	1 22			:	41	2 5	2 2	2 3	5	100				2 03	•	۸n. ۲	77.		3.87	3.89	0.00	2.52		4.13	1.76	1.40		1.75	3.23	1.76	6.75
Species		SrSiO3	Ğ	A1203	SrO		SrSiO3	Heb	1205	Sro	:	Section	Mosion	41211	60.3	7	5.0:01	10014	200	;	Casioi	4	30214	ged		CaSiO3	MgSiO3	A1203	SioZ		Srsio3	A1203	\$r0		K25i03	MgSiO3	A1203	2015
Moles	in 100g	967.0	0.030	0.239	0.250		0.401	0, 122	W 178	0.138	!	A NA	212	010	3,44		0 701	101	171	:	0.293	001	6.426	0.324		0.387	0.389	0.008	0.252		0.413	0.176	0.140		0.175	0.323	0.176	0.377
Holes		9.5.0	0.000	0.239	0.296		0.539	0.122	971.0	0.401		918	0.310	010	602		C75 U	101	0.401		0.617	001	0.426	0.293		0.387	0.389	0.000	1.028		0.553	0.176	0.413		0.175	0.323	0.176	0.675
	Rational ized	56.6	1.2	54.4	17.8	100.0	55.9	6.7	15.1	24.1	100.0	32.9	12.5	1.0	53.6	101	2.95	19.7	24.1	100.0	34.6	7.7	43.4	17.6	100.0	21.7	15.7	8.0	61.8	100.0	57.3	17.9	24.8	100.0	16.5	13.0	17.9	.52.6
5	j	55.0	1.2	23.7	17.3	97.2	55.0	8.9	14.0	23.7	98.3	33.0	12.5	1.0	53.8	160.3	26.0	16.7	24.0	99.7	35.0	5.7	74.0	17.8	101.3	21.7	15.7	8.0	8.19	0.00	\$6.5	17.6	7.72	98.5	16.5	13.0	18.0	\$2. <b>8</b>
5	Oxide	5r0	N90	A1203	2015	lotal	Sro	05H	A1203	2018	letal	Sro	06#	A1203	2018	lotal	510	A1205	2015	lotal	CaO	OPH	A1203	2015	lotal	Ca O	Og M	A1203	2015	letel	Sro	A1203	Si02	lotal	K20	æ æ	A1203	Si02
100		SHAS(A)		- ::			SHAS(B)					SHS(A)					SAS(C)				CMAS(A)					SMA2					SAS(0)				KHAS(A)	_		

### TABLE 12

	G <sub>hyd</sub>	
Relevant oxides	(kcal/mole)	(kJ/mol)
$SiO_2 + H_2O \longrightarrow H_2SiO_3$ (vitreous silica)	5.6	23.4
$Al_2O_3 + 3H_2O \longrightarrow 2Al(OH)_3$	3.1	13.0
$MgO + H_2O \longrightarrow Mg(OH)_2$	<del>-</del> 6.5	-27.2
$CaO + H_2O \longrightarrow Ca(OH)_2$	-13.1	-54.8
$sro + H2O \longrightarrow Sr(OH)2$	-17.3	-72.4
$\text{Na}_2\text{O} + \text{H}_2\text{O} \longrightarrow \text{2NaOH}$	-33.5	-140.2
$K_2O + H_2O \longrightarrow 2KOH$	-46.1	-192.9
$Tio_2 + H_2O \longrightarrow Ti(OH)_2O$	16.0	66.9
$P_2O_5 + 3H_2O \longrightarrow 2H_3PO_4$	-55.9	-233.9
$B_2O_3 + 3H_2O \longrightarrow 2H_3BO_3$	-9.8	-41.0
$zro_2 + H_2O \longrightarrow zr(OH)_2O$	-7.1	-29.7
Relevant Silicates		
$\text{Na}_2\text{SiO}_3 + 2\text{H}^+(\text{aq}) \longrightarrow \text{H}_2\text{SiO}_3 + 2\text{Na}^+(\text{aq})$	-28.8	-120.5
$K_2SiO_3 + 2H^+(aq) \longrightarrow H_2SiO_3 + 2K^+(aq)$		-174.5
$MgSiO_3 + 2H^+(aq) \longrightarrow H_2SiO_3 + Mg^{2+}(aq)$		-58.2
$Casio_3 +2H^+(aq) \longrightarrow H_2SiO_3+Ca^{2+}(aq)$		-67.4
$SrSiO_3 + 2H^+(aq) \longrightarrow H_2SiO_3 + Sr^{2+}(aq)$		-102.1
$Basio_3 + 2H^+(aq) \longrightarrow H_2Sio_3 + Ba^{2+}(aq)$	-37.3	-156.1
$ZnSiO_3 + 2H^+(aq) \longrightarrow H_2SiO_3 + Zn^{2+}(aq)$	-2.4	-10.0
<u>Disilicates</u>		
$Ca_2SiO_4+3H^+(aq) \longrightarrow H_2SiO_3+2Ca^{2+}+OH^-$	-30.3	-126.8
$Sr_2Sio_4 + 3H^+(aq) \longrightarrow H_2Sio_3 + 2Sr^{2+} + OH^-$		-210.5
$Mg_2SiO_4 + 3H^+(aq) \longrightarrow H_2SiO_3 + 2Mg^2 + + OH^-$	-17.9	-74.9

Although calculations were undertaken assuming the presence of the simplest silicates choice of other silicates (such as disilicates) does not appear to change the calculations much. For example given below is the calculation for an assumed composition which shows only minor differences for the free energy of hydration calculated.

```
0.767 CaO
Composition (moles/100g)
                                     0 417 MgO
                                     0.669 SiO2
0.384 \text{ Ca}_2\text{SiO}_4 =
                        -11.6
                                     or 0.433 CaSiO<sub>3</sub>
                                                                    -7.0
0.285 MgSiO3
                                           0.236 MgSiO3
                         - 4.0
                                                                    -3.3
0.132 MgO
                         <u>-0.9</u>
                                           0.334 CaO
                                                                    -4.4
                         -16.5
                                                                    -1.2
                                           0.181 MgO
                                                                   -15.9
                                         0.384 \text{ Ca}_{2}\text{SiO}_{4} =
                                     or
                                                                    -11.6
                                           0.132 \text{ Mg}_2 \text{SiO}_4 =
                                                                    - 2.4
                                           0.153 MgSiO<sub>3</sub>
                                                                    - 2.1
                                                                    -16.1
```

The applicants have found that when the free energy of hydration is more negative than -10kcal/100 grams (-418.4kJ/kg) of composition the composition showed high solubility. The compositions where this relationship broke down were those for which the total solubility was not available (for example those materials containing sodium, where any dissolved sodium would be swamped by the sodium in the saline solution) or where the free energy of hydration of the most likely species present was not available from the literature.

As a test of this technique the two examples of European Patent No. 0399320 were examined. The disclosed examples had the compositions:-

Component	Composition 1	Composition 2
	Weight percent	Weight percent
SiO <sub>2</sub>	60.7	58.5
Al <sub>2</sub> 0 <sub>3</sub>	-	5.8
CaO	16.5	3.0
Mg0	3.2	-
B <sub>2</sub> O <sub>3</sub>	3.3	11.0
Na <sub>2</sub> O	15.4	9.8
κ <sub>2</sub> ο	0.7	2.9
Iron oxide	0.2	0.1
BaO	-	5.0
ZnO	-	3.9

Using the above method of calculation Composition 1 had a free energy of hydration of -11.6kcal/100grams (-485.3kJ/kg) whereas Composition 2 had a free energy of hydration of -5.8kcal/100grams (-242.6kJ/kg). This would suggest that Composition 1 would be a saline soluble fibre, and hence physiologically safer than an insoluble fibre; whereas Composition 2 would be predicted to be a relatively insoluble fibre and hence less safe. This is what is disclosed in EP 0399320, the fibres of Composition 2 having a longer lifetime introduced fibres were which the studies in in interperitoneally into rats.

As mentioned above this predictive test can fail under some circumstances. To avoid these difficulties the applicants looked to a different predictive technique, namely the assessment of the amount of non-bridging oxygens present. This is calculated by normalising the chemical analysis to 100 weight%; calculating the molar percentage of each oxide; summing the oxygen-weighted contribution of each oxide to get the total number of oxygens; summing the weighted contribution of each oxide of non-bridging oxygens (see below); and taking the ratio of non-bridging oxygens to the total number of oxygens. The applicants have found that when this figure exceeds 30% the fibres are soluble.

To explain the term non-bridging oxygen one must look to the structure of glasses. Glasses are very stiff liquids and to form usually require the presence of a material that can form a network (usually an oxygen-bridged network). The network may be modified by constituents that contribute non-bridging parts to the network and open the structure of the network and so prevent crystallisation. These materials are usually referred to as network-formers and modifiers respectively.

The terms modifier and network former are well known in the glass industries. Network formers are materials such as  $SiO_2$ ,  $P_2O_5$ ,  $B_2O_3$  and  $GeO_2$  which can form an interconnected network to form the glassy phase. Modifiers are substances such as CaO,  $Na_2O$ , and  $K_2O$  which alter the network and have effects on such properties as viscosity and melting point. There are some intermediate materials (such as  $Al_2O_3$ ,  $TiO_2$ , PbO, ZnO and BeO) which can act as both network formers and modifiers depending on the environment and on the amount present.

In the above mentioned test, for calculating the non-bridging oxygens, one ignores the network formers and calculates the contribution of each other oxide. The contribution of each oxide depends on the geometry and charge of each cation in the glass. As examples typical contributions are as follows:-

- Ca<sup>2+</sup>,Mg<sup>2+</sup>,Sr<sup>2+</sup> and other divalent network modifier cations contribute 2 non-bridging oxygens
  - ${ t K}^+, { t Na}^+$  and other monovalent network modifier cations contribute 1 non-bridging oxygen
  - ${\rm Al}^{3+}, {\rm Ti}^{3+}$  and other intermediate cations contribute -1 non-bridging oxygen (i.e these oxides <u>reduce</u> the number of non-bridging oxygens)
  - $(\mathrm{Ti}^{4+}\ \mathrm{is}\ \mathrm{reduced}\ \mathrm{to}\ \mathrm{Ti}^{3+}\ \mathrm{in}\ \mathrm{most}\ \mathrm{glasses}\ \mathrm{when}\ \mathrm{present}\ \mathrm{in}\ \mathrm{relatively}\ \mathrm{small}\ \mathrm{quantities})$

38
TABLE 13

_					_					_											1			_				1			-								ì			_
Log(norm	solub)	1.63				1.82				1 88	3			1.84				2.40			2.21			09.0				06.0				0.78			00 0	2			2.45			
Normal ized	Solubillity	43				33			•	76	2			69				549			163			7				8				9			8	,			280			
S.S.A	#12/g	0.30				0.39				0.36	}			0.40				0.37			0.45			0.56				0.62				0.52			05.0	?			0.41			
lity	(bbu)	77	•	7	51	58	-	77	103	55	•	55	110	62	0	67	Ξ	2	367	369	1.7	253	762	*	4	\$	6	:	4	<u>د</u>	20	~	~	<b>~</b> :	2 0	. ^	· v	91	75	30	355	760
Solubility	0xide	CaO	A1203	Sio2	fotal	Ca0	A1203	Si02	Total	Ca0	A1203	Si02	Total	Ca0	1102	Si02	Total	SrO	2018	Iotal	Ca0	2015	lotal	Na20	A1203	Sio2	Total	CaO	A1203	Sio2	Total	₩30	A1203	Si02	10191	. ¥1204	Si02	Total	SrO	Ca0	Si02	lotal
0.8.NX		8.87				7.87				50.4				55.8				8.89			64.2			-5.5				1.9				-5.4			7 1.	:			65.8			
N.B.O.		78.4				8.77				80.4				85.1				102.4			97.2			-12.6				3.8				-11.2			7 6	:			0.0%			
Oxygen	Total	160.8				160.9				159.6				152.5				148.8			151.4			228.3				1.861				505.6			7 100				150.5			
Hol.X		6.95	15.7	29.4	100.0	24.0	15.1	30.8	100.0	55.4	15.2	29.3	100.0	47.5	6.6	42.6	100.0	51.2	48.8	100.0	48.6	51.4	100.0	22.0	28.3	1.67	100.0	25.7	23.8	50.5	100.0	17.0	22.6	<b>7</b> .09	23.0	: K	20.8	100.0	23.5	26.0	50.5	100.0
Motes		0.842	0.2.0	0.451	1.533	0.829	0.232	0.473	1.534	0.852	0.234	0.451	1.537	192.0	0.159	0.682	1.602	609.0	0.581	1.190	0.822	0.870	1.692	0.303	0.389	0.684	1.376	0.374	0.347	0.736	1.457	0.248	0.331	0.882	0 28A	305	0.612	1.205	0.338	0.374	0.726	1.438
ition	×. =	47.2	24.5	27.1	Total	46.5	23.7	28.4	Iotat	8.72	23.9	27.1	Total	42.7	12.7	41.0	Total	63.1	34.9	Iotal	46.1	52.3	lotal	18.8	39.7	41.1	Total	21.0	35.4	7.75	Total	0.0	33.7	53.0	20.8	1 12	36.8	lotal	35.0	21.0	43.6	Total
Composition	Oxide	g Q	A1203	Si02		Oe J	A1203	Si02		CaO	A1203	Si02		CaO	1 102	\$ i02		Sr0	Si02		Ca0	S i 02		Na20	A1203	Si02		Ca0	A1203	Si02		06¥	A1203	\$ i05	0.50	A1203	5102		Sro	Ca0	2018	
Helt Code	١	CAS10(B) - A				CAS10(8) · B				CAS10(B) · C				C1S(A)				SrSi03			CaSi03			MAS(A)				CAS4				HAS(B)			(4/545	```			SCS(A)			

TABLE13 (continued)

CHACCAN				4.	10840	.0.0.	0.8.X	Solub	Solubility	4 3 3		
CALCANO	ox i de	H. X			lotal			Oxide	(mxx)		morning 1 2 ed	Log(norm
	Sro	55.0	0.531	1.67	169.6	8.09	35.8	Sco	-,25	6/70	2010011117	(dr) los
	<b>1</b>	1.2	0.030	2.8				3	; •	ġ.	\$2	1.58
	A1203	23.7	0.232	21.5				A1203	۰ ۸			
	Si02	17.3	0.288	56.6				\$102	· ^			
20/3/10/3	1	lotal	1.081	100.0				Total	\$9			
SHAS(B)	50	55.0	0.531	9.77	157.6	84.8	53.8	SrO	110	0 30	90	
	<b>W</b> 30	8.4	0.119	10.0				M <sub>Q</sub> 0	5	;	2	. ya
	A1203	8.7	0.145	12.2				A1203	· -			
	Si02	23.7	0.394	33.2				Sin	۲ -			
		Total	1.189	100.0				Toral	77			
SHS(A)	Sro	33.0	0.318	20.7	159.5	80.6	50.5	023	2/7	92.0		
	MgO	12.5	0.310	20.2		!	:		<u>?</u> ;	0.39	212	2.33
	A1203	1.0	0.010	9.0				2001	ē •			
	Si02	53.8	0.895	58.4				6013	0 0			
		Total	1.533	100.0				Total	C21			
SAS(C)	Sro	96.0	0.540	47.7	169.3	61.4	36.3	0.5	200	02.0	1	
	A1203	19.7	0.193	17.0			:	2021	₹ •	0.38	7<	1.72
	Si02	24.0	0.399	35.3				Sio	- «			
		Total	1.132	100.0				lotal	۶ ۾			
CHAS(A)	CaO	35.0	0.624	42.6	179.3	41.4	23.1	CaO		02.0	2,6	
	H <sub>9</sub> 0	4.5	0.112	7.6				ObM			į	1.38
	A1203	64.0	0.432	29.5			-	A1203	м			
	Si02	17.8	0.296	20.3				\$102	. 60			
		lotal	1.464	100.0				Total	8			
SWAZ	ල :	21.7	0.387	21.4	157.5	85.0	54.0	CaO	53	0.48	011	2 08
	064	7.51	0.389	21.5				MgO	25		:	2
	A1203	æ :	0.008	<b>7</b> .0;				A1203	0			
	7010	0.10	1.028	7.96.				Sio2	118			
SAS(D)	Sro	56.5	275 0	0.00.	0 776	6 77	. 4.	Total	228			
	A1203	17.6	0.173	15.4	ì	3.55	37.1	Sro.	ò.	0.40	85	1.96
	Si02	24.4	0.406	36.1				A1203	- %		-	
		Total	1.124	100.0				Total	771			
KHAS(A)	K20	16.5	0.175	11.3	179.5	41.4	23.1	K20		72.0	*	30.0
	MgO	13.0	0.323	20.8				001	. 00	3	:	?
	A1203	18.0	0.177	11.4				A1203				
	S i 02	52.8	0.879	9.95				S i 02	. =			
		lotal	1.554	100.0				Iotal	02			

Table 13 shows, for the compositions of Table 11, the calculated figure for non-bonding oxygens and these figures are reproduced in Fig.5. It can be seen that the plot of Fig.5 is more linear than that of Fig.4. Compositions with greater than 30% non-bridging oxygens show high solubility.

To illustrate the method the calculation for one example (the first shown in Table 13) is given below:-

	Mols/	Total Oxygen	Non-bridging Oxygens
CaO Al <sub>2</sub> O <sub>3</sub> SiO <sub>2</sub>	0.549 0.157 0.294	0.549 0.471 (0.157*3) 0.588	1.098 (2*0.549) -0.314 (-1*0.157*2Al) 0.000
	1.000	1.608	0.784

% non-bridging oxygens (0.784/1.608)\*100 = 48.8%

As an example one can look to European Patent Specification No. 0399320 referred to above. Using this method in relation to that specification Composition 1 has a non-bridging oxygen percentage of 48.2% whereas Composition 2 has a non-bridging oxygen percentage of 19.6%, again predicting that Composition 1 is more soluble than Composition 2.

There is a further criterion which a composition must meet for it to form vitreous fibres, namely that it be capable of forming a glass. The applicants have found a simple test. If the ratio of modifiers to network formers is less than a critical value (for SiO<sub>2</sub> based glasses, 1) the composition will generally form a glass. For the purpose of this test reasonable results are obtained if such intermediate materials are treated as networkers. Table 14 shows for a series of compositions in each column:-

Fibre reference (N.B. these are not the same fibres as shown in tables 9 & 10)

Composition

Molar ratios

Ratio of glass modifiers to network formers

Free energy of hydration of raw oxides

Melting Point

X-ray diffraction results

Solubility data (# indicates not measured)

Specific surface area

Normalised solubility data

Arbitrary ranking as to solubility and glass forming ability

Indication as to whether melting point above 1500°C

It should be emphasised that this test is a screening one rather than a wholly predictive one as there are several circumstances that may lead to its failure. Among these circumstances are compound formation and inability to quench fast enough to form glass.

Having adopted these tests as a screening method there follows a further step to ascertain whether the composition will form a vitreous fibre. This last step is best examined experimentally as fibre forming ability is a complex function of many physical characteristics, e.g. viscosity, which are often difficult to measure.

### Key to Fig.1

CR Cristobalite

TR Tridymite

PS Pseudowollastonite

WO Wollastonite

RA Rankinite

LI Lime

PE Periclase

FO Forsterite

PR Protoenstatite

DI Diopside

AK Akermanite

ME Merwinite

MO Monticellite

TABLE 14

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### CLAIMS

1. A method of predicting whether a given composition will form a saline soluble vitreous material, the method comprising the steps of:-

### EITHER

- a) assuming the composition to be a mixture of the compounds that would be present at equilibrium (on the basis of knowledge, informed belief or reasonable assumption);
- b) calculating the sum of the free energies of hydration of each of the compounds assumed to be present;
- c) if the sum of the free energies of hydration is more negative than -10 kcal/100grams (-418.4kJ/kg) of composition, assuming the composition to be soluble in saline solution;

OR

- a") calculating the percentage of non-bridging oxygens present in relation to the total oxygen present and, if the percentage is greater than 30% assuming the composition to be soluble in saline solution;
- 2. A method as claimed in claim 1 which further comprises in combination with steps a)-c) and/or a"):-
- d) calculating the ratio of glass modifier to network former present;
- e) if the ratio is less than a critical value (for SiO<sub>2</sub> based compositions, 1) assuming that the composition will form a vitreous material.

- 3. A saline soluble vitreous inorganic fibre selected and used in the knowledge that it has a composition meeting the criteria that:-
- the calculated sum of the free energies of hydration of the compounds that would or could be present at equilibrium (on the basis of knowledge, informed belief or reasonable assumption) is more negative than -10 kcal/100 grams (-418.4kJ/kg) of composition;

and/or

- b) the percentage of non-bridging oxygens present in relation to the total oxygen present is greater than 30%.
- 4. A saline soluble vitreous inorganic fibre as claimed in claim 3 selected and used in the knowledge that it has a composition meeting the criterion that the ratio of glass modifiers to network formers is less than a critical value (for SiO<sub>2</sub> based compositions, 1).
- 5. Use of a saline soluble vitreous inorganic fibre in the knowledge that it has a composition meeting the criteria that:-
- a) the calculated sum of the free energies of hydration of the compounds that would or could be present at equilibrium on the basis of knowledge, informed belief or reasonable assumption is more negative than -10 kcal/100grams (-418.4kJ/kg) of composition;

and/or

- b) the calculated percentage of non-bridging oxygens in relation to the total oxygen content is more than 30%.
- 6. Use as claimed in claim 5 of a saline soluble vitreous inorganic fibre in the knowledge that it has a composition meeting the criterion that the ratio of glass modifiers to network formers is less than a critical value (for SiO<sub>2</sub> based compositions, 1).

7. Use as saline soluble fibres having a shrinkage of less than 3.5% when exposed to 1000°C for 24 hours and/or 800°C for 24 hours, of vitreous fibres having a composition comprising (in weight %):-

```
5i0_2 >58% - (for MgO =< 10%) and 5i0_2 >58% + 0.5(%MgO -10) - (for MgO >= 10%) CaO 0% - 42% MgO 0% - 31.33% Al<sub>2</sub>O<sub>3</sub> 0% - <3.97%
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and being essentially free of fluxing components such as alkali metals and boron oxide.

- 8. Use as claimed in claim 7 of saline soluble fibres having a shrinkage of less than 3.5% when exposed to 1000°C and/or 800°C for 24 hours, of vitreous fibres in which the amount of SiO<sub>2</sub> is <70%.
- 9. Use as claimed in any of claims 7-8 of saline soluble fibres having a shrinkage of less than 3.5% when exposed to 1000°C and/or 800°C for 24 hours, the fibres crystallising as diopside and having the composition consisting essentially of:-

Component	Composition A
	Weight percent
SiO <sub>2</sub>	59-64
Al <sub>2</sub> O <sub>3</sub>	0-3.5
CaO	19-23
MgO	14-17

10. Use as claimed in any of claims 7-8 of saline soluble fibres having a shrinkage of less than 3.5% when exposed to 1000°C and/or 800°C for 24 hours, the fibres crystallising as wollastonite and/or pseudowollastonite and having the composition consisting essentially of:-

Component	Composition B
	Weight percent
sio <sub>2</sub>	60-67
Al <sub>2</sub> O <sub>3</sub>	0-3.5
CaO	26-35
MgO	4-6

- 11. Use as a saline soluble fibre as claimed in any of claims 5-10 and in which, owing to the increase in molar volume on crystallisation, the linear shrinkage at the maximum service temperature is less than 3.5%.
- 12. A method for the manufacture of refractory oxide fibres containing calcium and silicon by the formation of an oxide melt containing calcium and silicon characterised in that all or part of the calcium and all or part of silicon is provided by a calcium silicate.
- 13. A method as claimed in claim 13 in which the calcium silicate is wollastonite.

### **ABSTRACT**

Disclosed is use of a vitreous inorganic fibre in the knowledge that it has a composition meeting the criterion that the calculated sum of the free energies of hydration of the compounds that would or could be present at equilibrium (on the basis of knowledge, informed belief or reasonable assumption) is more negative than -10 kcal/100grams of composition. Such compositions are saline soluble.

### IN THE UNITED STATES PATENT AND TRADEMARK OFFICE



Applicants:

Gary Anthony Jubb and Jean-Louis Martin

Serial No.:

Divisional of 09/262,378

Filing Date:

October 24, 2000

For:

SALINE SOLUBLE INORGANIC FIBRES

Box Patent Application

Date: October 24, 2000

Director of Patents and Trademarks

Washington, D.C. 20231

### TRANSMITTAL OF FORMAL DRAWINGS

Sir:

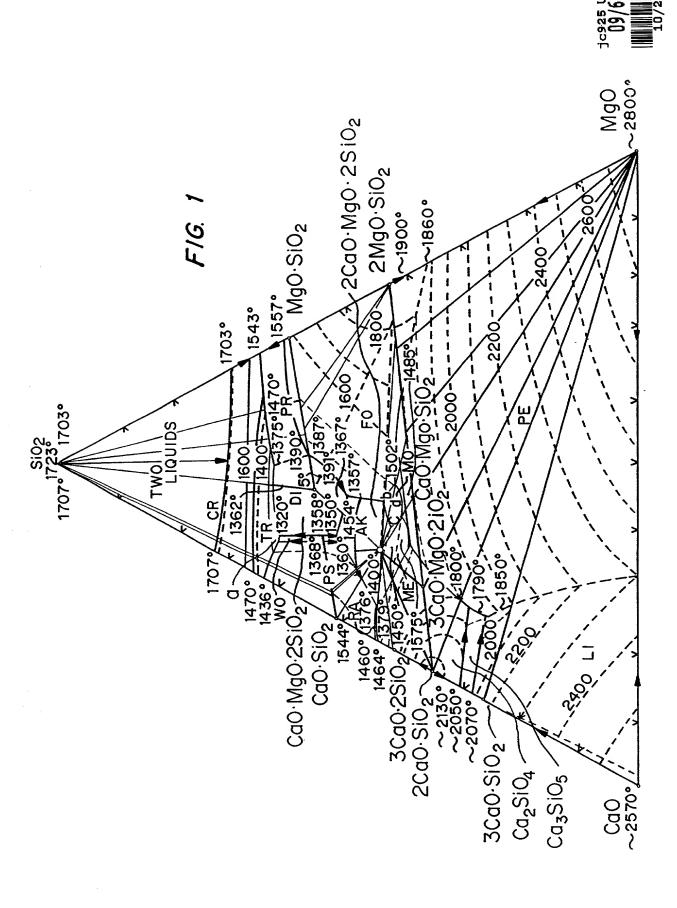
Applicants transmit herewith three sheets of Formal Drawings in connection with the filing of this divisional application. Applicants paid the issue fee in parent U.S. Serial No. 09/262,378 on September 29, 2000, and at that time provided the U.S. Patent Office with identical Formal Drawings.

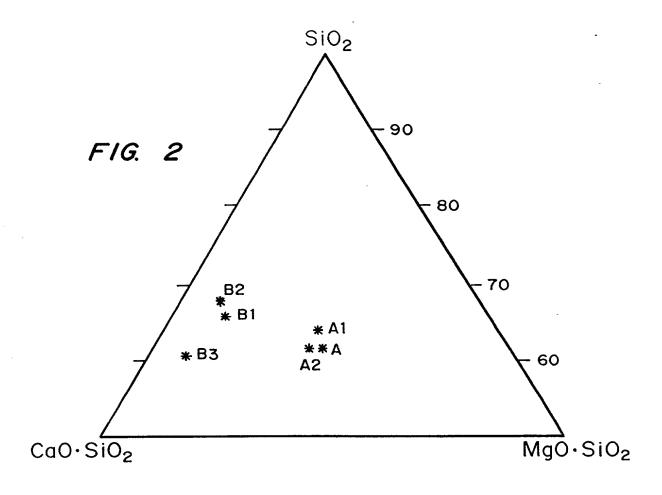
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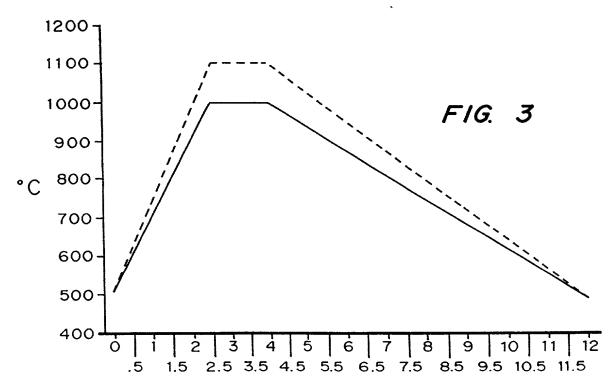
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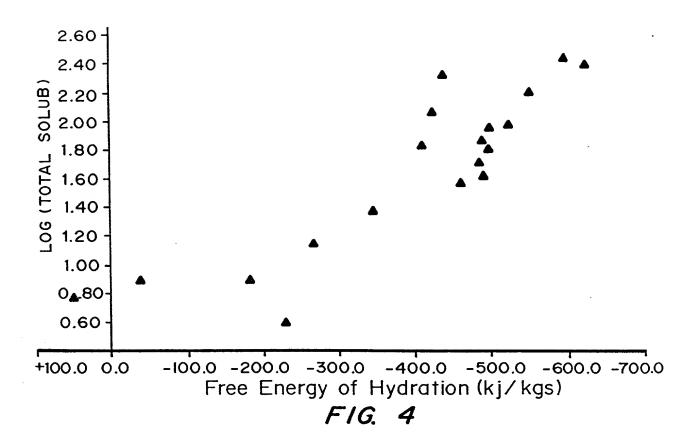
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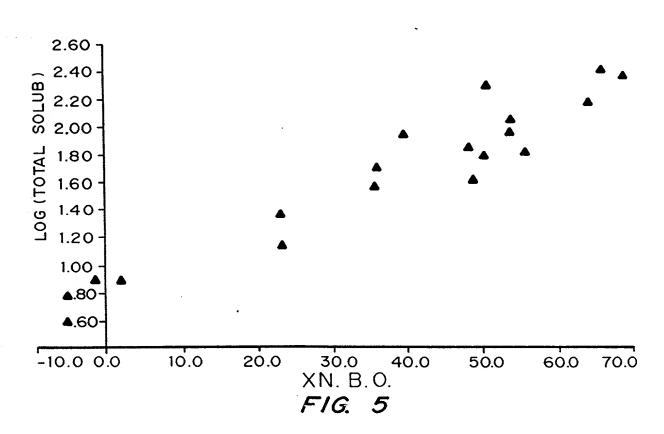
OF COUNSEL:
KILPATRICK STOCKTON LLP
Suite 2800
1100 Peachtree Street
Atlanta, Georgia, 30309-4530
404-815-6218
Attorney Docket No.: M8540/248465











COMBINED DECLARATI (Industry Sections to PCI Inter	MBINED DECLARATION FOR PATENT APPLICATION AND POWER OF ATTORNEY  MB540/188317							
ža zĄ	elow named inventor, I bereby de	dare that:						
My residence,	My residence, post uffice address and ciriosnehip are as stated below next to my name.							
Some miserial	he original, first and sole inventor If playal names are listed below) of invention entitled:	(if only one name is listed below the subject matter which is cla	) or en original, first imed and for which a	and patent				
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	I hereby claim foreign priority henefits under Title 25, United State code, §119 of any foreign application(s) for patent or inventor's cartificate or of any PCT international application(s) designating at least one country other than the United States of America listed below any foreign application(s) for patent or inventor's continue and BCT international application of the patent or inventor's continue and BCT internation of the patent of the							
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PRIOR FOREIGN/PCT APPLICATION(S) AND ANY PRIORITY CLAIMS UNDER 35 U.S.C. 119:								
COUNTRY	APPLICATION NUMBER	DATE OF FILING	PRIORITY CLAD UNIORE SE UNC	air				
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(Combined Declaration For Patent Application and Power of Attorney -- PTO 1351 [13-11]-page 1 of 2)

903D93 COMBRUED DECLARATED ME PADENT APPLICATION AND POWER OF ATTORNEY

All Control of the Control COMBRES DECLARATION FOR PATRIT APPLICATION AND FOWER OF ATTORNEY (CONTINUED) (Includes Reference to PCT International Applications) M8540/188317

I hereby claim the benefit under Title 35, United States Code, §120 of any United States application(s) or PCT international application(s) designating the United States of America that is/are listed below and, insofar as the subject matter of each of the claims of this application is not disclosed in that/those prior application(s) in the manner provided by the first paragraph of Title 35, United States Code, §112, I acknowledge the duty to disclose material information as defined in Title 37, Code of Federal Regulations, §1.56(a) which occurred between the filing date of the prior application(s) and the national or PCT international filing date of this application:

PRIOR U.S. APPLICATIONS OR PCT INTERNATIONAL APPLICATIONS DESIGNATING THE U.S. FOR BENEFIT UNDER 35 U.S.C. 120:

<del>-</del>	U.S. APPLICATIONS			STATUS (Check One)		
U.S. APPLIC NUMBE		U.S. FILING DATE	PATENTED	PENDING	ABANDONED	
08/535,587		28 September 1995 (28.69.95)			T .	
<b>0</b> 87899, <b>0</b>	<b>\$</b> 5	23 July 1997 (23.07.97)		×		
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APPLICATION NO	PCT FILING DATE	U.S. SERIAL NUMBERS ASSECTED (If my)				
PCT/GR93/ 00085	15 January 1993 (15.01.93)	08/039,486			*	
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POWER OF ATTORNEY: As a sensed inventor, I hereby appoint the following attorney(s) and/or agent(s) to prosecute this application and transact all business in the Patent and Trademark Office connected therewith. (List name and registration number) Charles Y. Lackey 22,707; John M. Harrington 25,592; John S. Prutt 29,476; A. Jose Cortina 29,733; James L. Ewing. IV 30,630; Charles W. Callons 31,814; George T. Marcon 33,814; Denn W. Russell 31,452; Richard T. Peterson 35,320; Charles T. Simmons 35,359; Nora M. Tocups 35,717; Bruce D. Gray 35,799; Theodore R. Harper 35,896; Kristin L. Johnson 44,807; Gooff L. Sutcliffe 36,348; Pat Winston Kennedy 36,570; Mitchell G. Stockwell 39,389; Michael J. Turton 40,252; Youthin L. Kundupoglu 41,130; Benjamin D. Driscoll 41,571; Wilhurn L. Chesser 41,668; Alama G. Kringunan 41,747; J. Steven Gardner 41,772; Junes J. Bindseil 42,326; Richard H. Lilley 42,803; Edwins Thomas Washington 43,187; Camilla Camp Williams 43,992; Carl B. Massey P44,224; R. Whitney Winston P44,432; John William Hall, Jr. P44,433

Send Correspondence to: John S. Pratt, Esq. Kilpatrick Stockton LLP 1100 Peachtree Street, Suite 2800 Adanta, GA 30309-4530				Direct Telephone Calls to:
2	OS BARKION BUILT NAVE	JUBB 1	GARY	ANTHONY
•	CLINESSESS. REMINERCY V	CON Bewdley	United Kingdom	U.K.
Ľ	POST OFFICE	1 Wynn Close,	Wribbenhall, Bewdley. DY12 1JR	STATE & ZIF CODE/COCHYRY
1	OF BOXEN FOR	MARTIN	JEAN-LOUIS	MENTE GIVEN NAME
•	COMMISSION N	Manthrison	France Prance	France
2	ADDRESS.	6 Impasse des Genets	Monthrison	France 42600
7	MANAGE CA	PARGLY RAME	STEET COVER NAME.	SECOND CLAIM NAME
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I hereby declare that all statements made herein of my own knowledge are true and that all statements made on information and belief are believed to be true, and further that these statements were made with the knowledge that willful fulse statements and the like so made are punishable by fine or imprisonment, or both, under section 1001 of Title 18 of the United States Code, and that such willful false statement may jeopardize the validity of the application or any patent issuing thereon.

DONA Gary Anthony July

Jean-Louis Martin

903093 COMBINED DECLARATION FOR PATENT APPLICATION AND POWER OF ATTOROGY

### IN THE UNITED STATES PATENT AND TRADEMARK OFFICE

APPLICANTS:

Gary Anthony Jubb and Jean-

Louis Martin

Continuation of

**GROUP ART** 

SERIAL NO.:

08/899,005

UNIT:

1755

**DEPOSIT DATE:** 

March 4, 1999

**EXAMINER:** 

Karl Group

FOR:

SALINE SOLUBLE INORGANIC FIBERS

Assistant Commissioner for Patents BOX PATENT APPLICATION

Attorney Docket No. 73405DIV

Washington, D.C. 20231

DATE: March 4, 1999

### APPOINTMENT OF ASSOCIATE ATTORNEY

Sir:

Please recognize Bruce D. Gray, Reg. No. 35,799 of the firm of Kilpatrick Stockton LLP, 1100 Peachtree Street, Suite 2800, Atlanta, GA 30309-4530 as an Associate Attorney in the above-entitled application.

Respectfully submitted,

Dean W. Russell

Reg. No. 33,452

KILPATRICK STOCKTON LLP 1100 Peachtree Street, Suite 2800 Atlanta, GA 30309-4530 404.815.6500

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		<del></del>		APPLICATION
NOTE:	If the application filed more that forming the basis for this applicantinuation, divisional, or coldectaration and POWER OF All of the prior U.S. or PCT application	ication entering the ntinuation-in-part, th TTORNEY FOR DIVISION	United States as ( nen also complete NAL, CONTINUATIOI	1) the national stage, or (2) a ADDED PAGES TO COMBINED
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and tran	by appoint the following a sact all business in the Particular in			
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	PRRESPONDENCE TO			ELEPHONE CALLS TO: e and telephone number)
Kilpa 1100 Suite	S. Pratt trick & Cody Peachtree Street 2800			W. Russell 815-6528
U.S.A	ta, Georgia 30309	M		
		DECLARATIO	ON	
	by declare that all statem statements made on infor se statements were made so made are punishable	mation and belice with the knowledge by fine or impris	ef are believed ledge that willf sonment, or bo willful false sta	to be true; and further ul false statements and oth, under Section 1001
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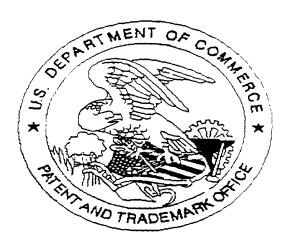
Post Office Address 16 Walton Close, Stourport-on-Severn, Worcestershire DY13 OLS, England

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(Rei 54-11/92 Pub 605)	FORM 1-1	1-11
(Declaration and Pov	ver of Attorney [1-1]—page	4 of 5)
Full name of second jo	oint inventor, if any <u>Jean-Louis</u>	Martin
Jean-Louis	NMI	Martin
(GIVEN NAME)	(MIDDLE INITIAL OR NAME)	FAMILY (OR LAST NAME)
Inventor's signature _ Date25/3/93	Country of Citizenshi	France
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<b>XDS/X</b> XXXXXXXXXXXXXXXX	<u> </u>	\$XXXXXXXXXXXXXXXXXXXXXXXXXXXXXXXXXXXXX
Residence Montbri	son, France	
Post Office Address _6	Impasse des Genets, 42600	Montbrison, France

## SCANNED. #

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